



# **High energy X-ray absorption spectroscopy:** **Current status and future applications**

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**Workshop on  
“Science with High-Energy X-rays”  
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# Overview

## **X-ray absorption**

- Principles of XANES und EXAFS
- Edge positions of K- and L-edges
- Lifetime broadening – and how to get rid of it

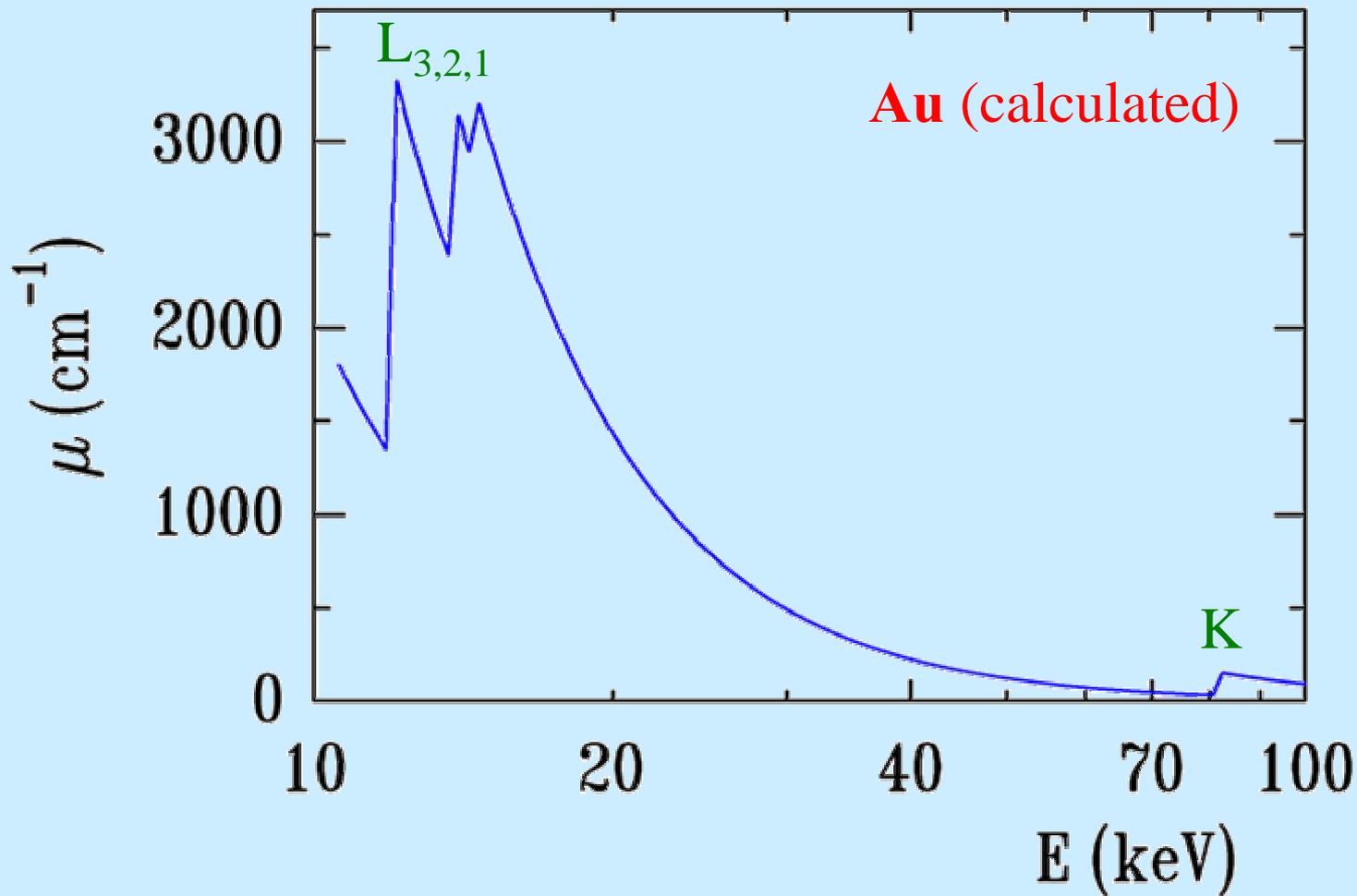
## **Applications**

- Metals
- Melting
- Time resolution
- Tomography
- Beamline layout at PETRA

## **Conclusions**



# X-ray absorption edges



**$^{79}\text{Au}$**

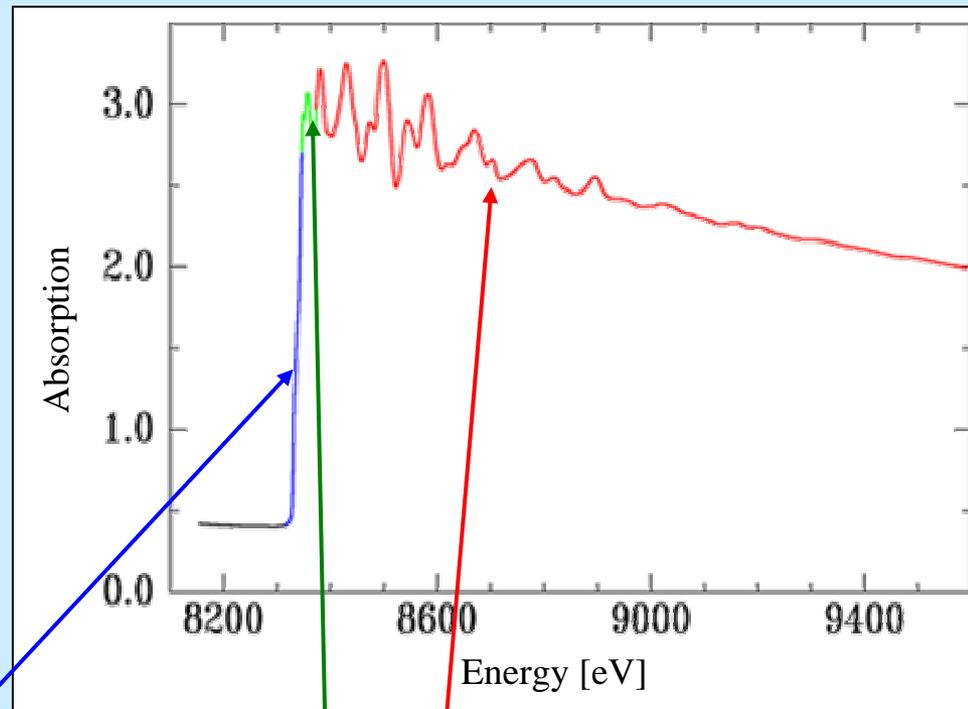
**$K$ : 80.725 keV**

**$L_3$ : 11.919 keV**

**$L_2$ : 13.734 keV**

**$L_1$ : 14.353 keV**

# Absorption spectra of a Ni-metal foil



**Absorption edge**

⇒ **Valence of the absorbing atom**

**Near edge region (XANES)**

⇒ **Bond angles**

⇒ **„Fingerprinting“**

**Extended edge region (EXAFS)**

⇒ **Interatomic distances**

⇒ **Coordination number**

⇒ **Identification of the neighbour atoms**



# Theoretical description of the EXAFS

K-edges, **single** gaussian pair distribution of neighboring atoms:

$$\chi(k) = -\frac{1}{k} A(k) \cdot \sin \left[ 2kr + \Phi(k) \right]$$

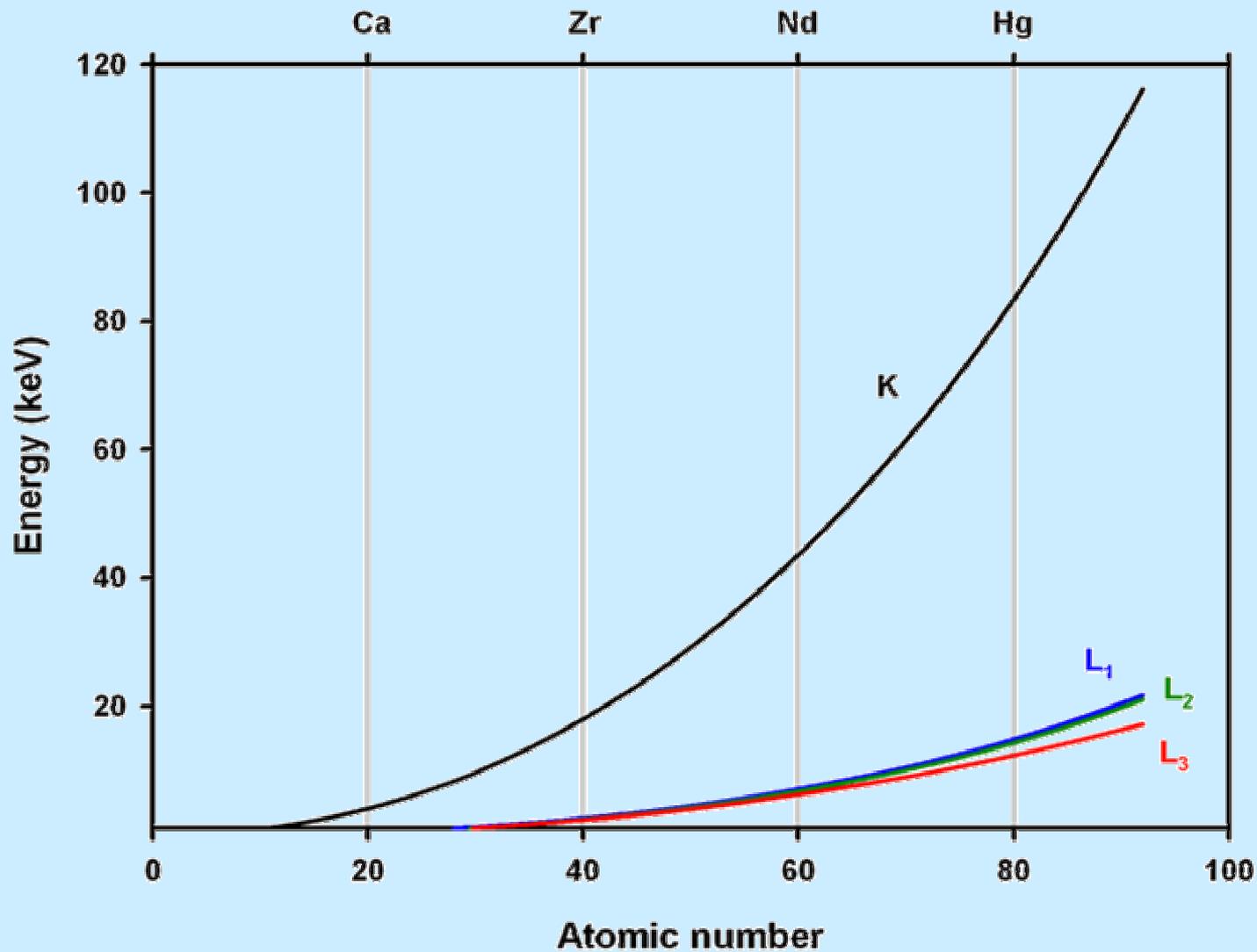
where

$$A(k) = \frac{N}{r^2} |f(\pi, k)| e^{-2r/\lambda} e^{-2\sigma^2 k^2}$$

<b>k</b>	<b>wavenumber of photoelectrons</b>
<b>r</b>	<b>average atomic distance</b>
<b><math>\phi(k)</math></b>	<b>scattering phase due to potentials of central atom (<math>\phi_A</math>) and backscattering atom (<math>\phi_B</math>), <math>\phi = \phi_A + \phi_B</math></b>
<b>N</b>	<b>coordination number</b>
<b><math> f(\pi, k) </math></b>	<b>backscattering amplitude</b>
<b><math>\lambda</math></b>	<b>mean free path of photoelectrons</b>
<b><math>\sigma</math></b>	<b>Debye-Waller-factor</b>



# Edge energies





# What are „high energy absorption edges“?

## Heaviest tabulated element:

${}_{100}\text{Fm}$	K	141.930 keV
	L <sub>1</sub>	27.573 keV
	L <sub>2</sub>	26.644 keV
	L <sub>3</sub>	20.868 keV

R.D. Deslattes et al.,  
Rev. Mod. Phys. 75, 35 (2003)

⇒ Above 30 keV only K-edges, >  ${}_{51}\text{Sb}$  ( $E_{\text{K}}=30.491$  keV)

## Advantages of K-edges:

- Only one (primary) final state (p-state)
- No overlap with other edges
- Polarization dependent measurements:  
Final state has simple p-symmetry



# Edge energies in eV

$^{57}\text{La}$  K 38 925     $^{72}\text{Hf}$  K 65 360     $^{29}\text{Cu}$  K 8 970     $^{79}\text{Au}$  K 80 725  
 $L_1$  6 266     $L_1$  11 271     $L_1$  1 096     $L_1$  14 353  
 $L_2$  5 891     $L_2$  10 739     $L_2$  952     $L_2$  13 734  
 $L_3$  5 483     $L_3$  9 561     $L_3$  932     $L_3$  11 919

$^{92}\text{U}$  K 115 606  
 $L_1$  21 757  
 $L_2$  20 948  
 $L_3$  17 166

1 H Hydrogen																	2 He Helium						
3 Li Lithium	4 Be Beryllium																	5 B Boron	6 C Carbon	7 N Nitrogen	8 O Oxygen	9 F Fluorine	10 Ne Neon
11 Na Sodium	12 Mg Magnesium																	13 Al Aluminum	14 Si Silicon	15 P Phosphorus	16 S Sulfur	17 Cl Chlorine	18 Ar Argon
19 K Potassium	20 Ca Calcium	21 Sc Scandium	22 Ti Titanium	23 V Vanadium	24 Cr Chromium	25 Mn Manganese	26 Fe Iron	27 Co Cobalt	28 Ni Nickel	29 Cu Copper	30 Zn Zinc	31 Ga Gallium	32 Ge Germanium	33 As Arsenic	34 Se Selenium	35 Br Bromine	36 Kr Krypton						
37 Rb Rubidium	38 Sr Strontium	39 Y Yttrium	40 Zr Zirconium	41 Nb Niobium	42 Mo Molybdenum	43 Tc Technetium	44 Ru Ruthenium	45 Rh Rhodium	46 Pd Palladium	47 Ag Silver	48 Cd Cadmium	49 In Indium	50 Sn Tin	51 Sb Antimony	52 Te Tellurium	53 I Iodine	54 Xe Xenon						
55 Cs Cesium	56 Ba Barium	57 La Lanthanum	72 Hf Hafnium	73 Ta Tantalum	74 W Tungsten	75 Re Rhenium	76 Os Osmium	77 Ir Iridium	78 Pt Platinum	79 Au Gold	80 Hg Mercury	81 Tl Thallium	82 Pb Lead	83 Bi Bismuth	84 Po Polonium	85 At Astatine	86 Rn Radon						
87 Fr Francium	88 Ra Radium	89 Ac Actinium	104 Rf Rutherfordium	105 Db Dubnium	106 Sg Seaborgium	107 Bh Bohrium	108 Hs Hassium	109 Mt Meitnerium	110	111	112												

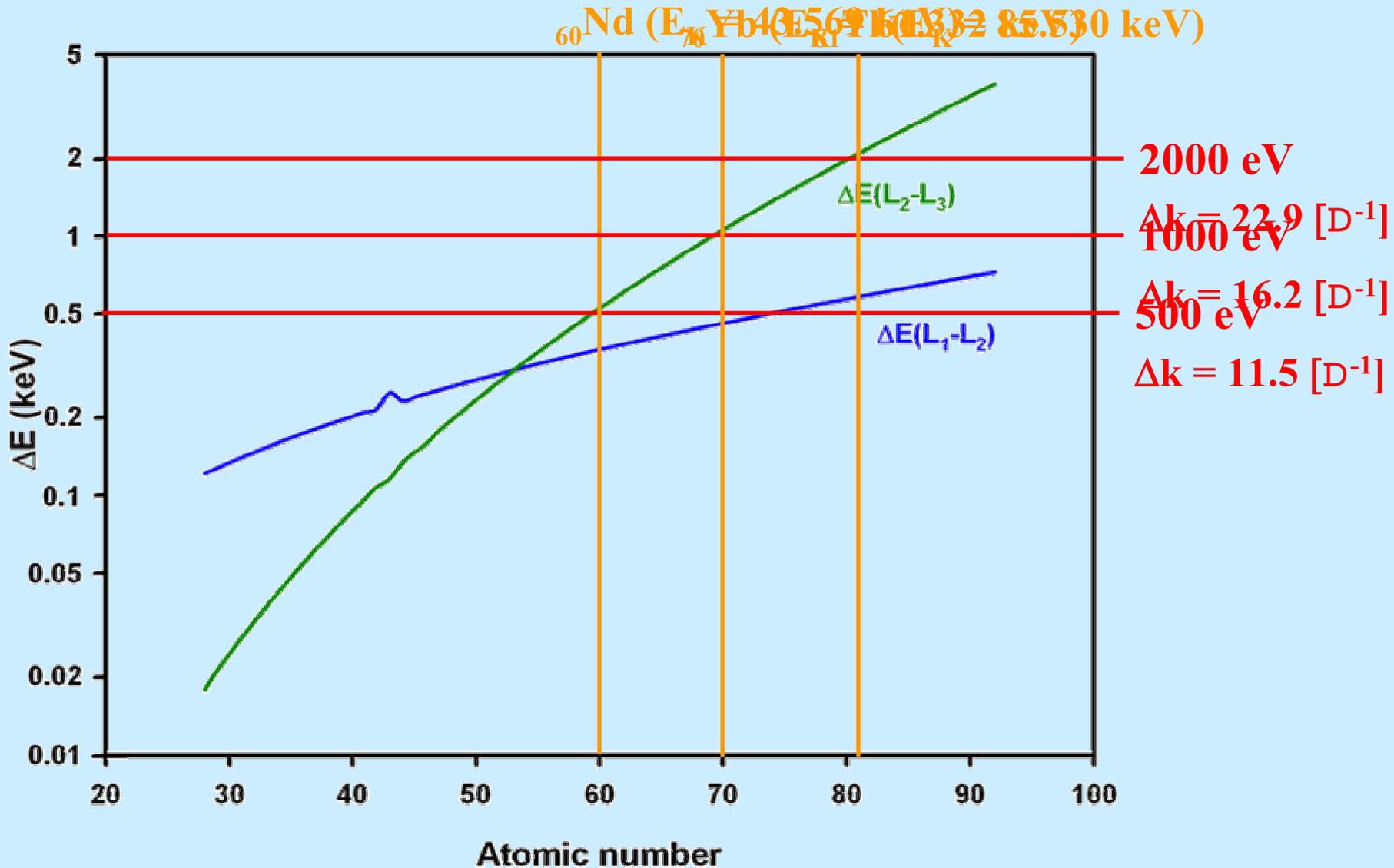
Lanthanide series →

58 Ce Cerium	59 Pr Praseodymium	60 Nd Neodymium	61 Pm Promethium	62 Sm Samarium	63 Eu Europium	64 Gd Gadolinium	65 Tb Terbium	66 Dy Dysprosium	67 Ho Holmium	68 Er Erbium	69 Tm Thulium	70 Yb Ytterbium	71 Lu Lutetium
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Actinide series →

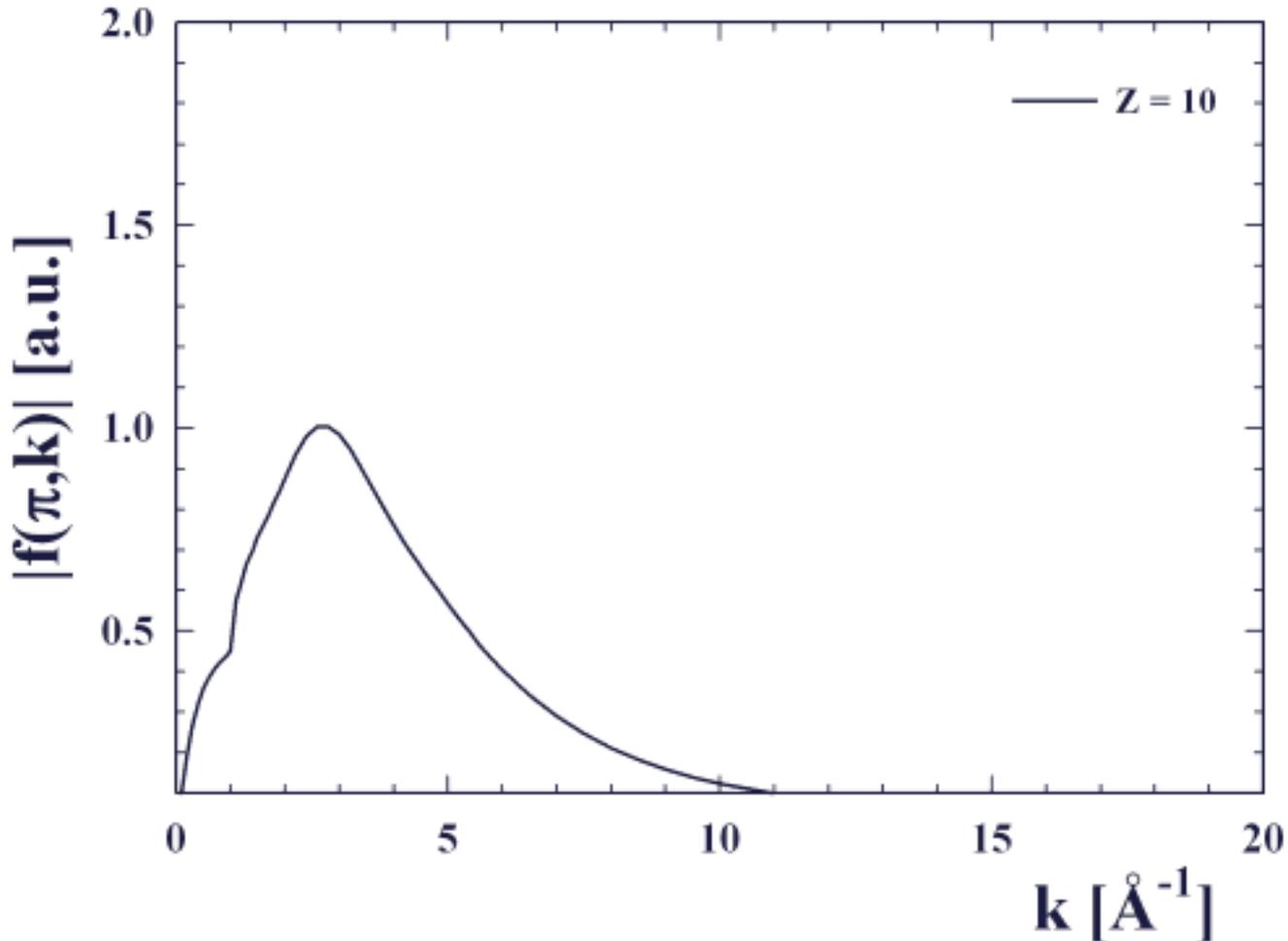
90 Th Thorium	91 Pa Protactinium	92 U Uranium	93 Np Neptunium	94 Pu Plutonium	95 Am Americium	96 Cm Curium	97 Bk Berkelium	98 Cf Californium	99 Es Einsteinium	100 Fm Fermium	101 Md Mendelevium	102 No Nobelium	103 Lr Lawrencium
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# EXAFS ranges at L<sub>3</sub>- and L<sub>2</sub>-edges





# Backscattering amplitudes

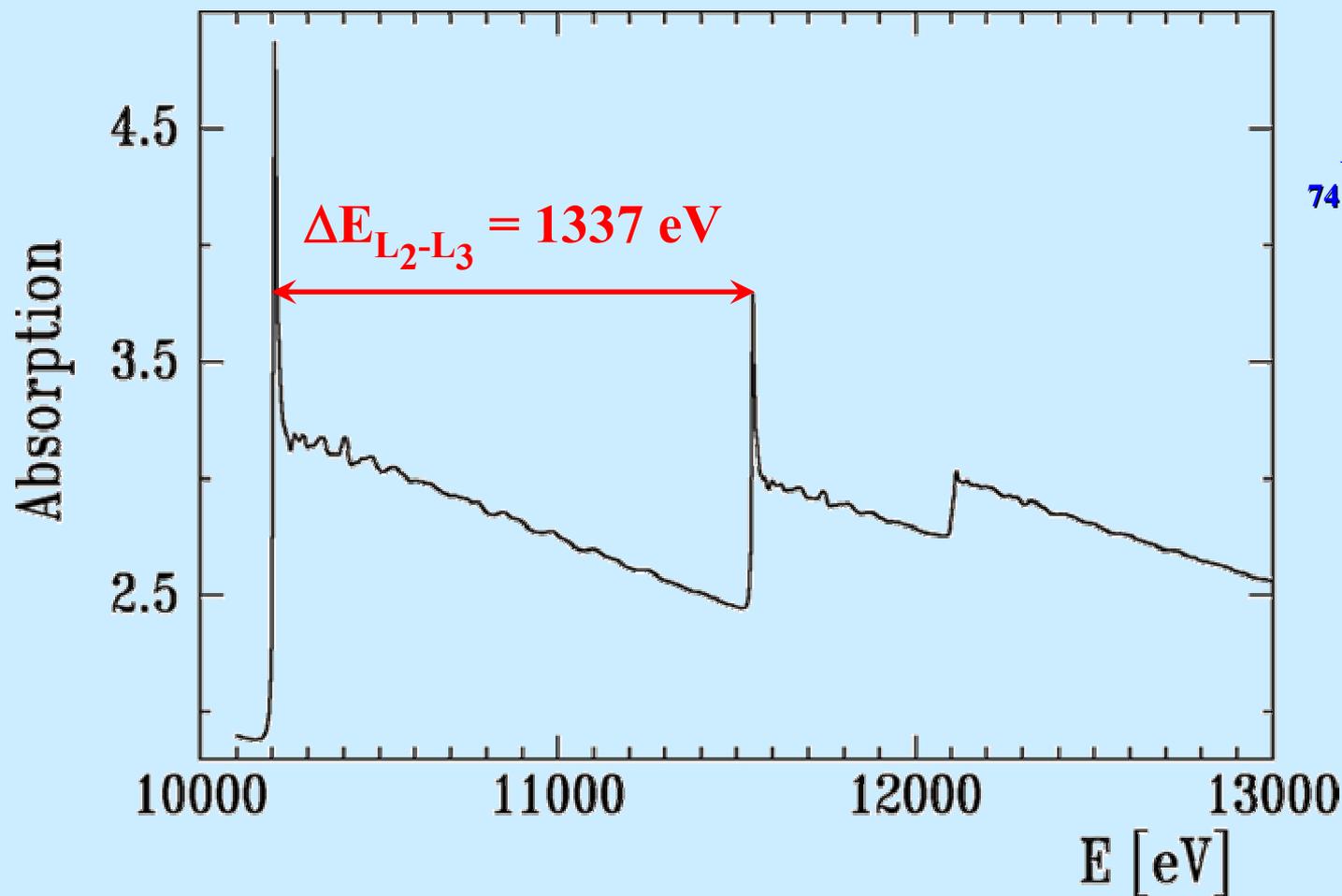


BSAs for  $Z = 10 - 90$

Calculated with FEFF-program



# Wide energy range: W L-edges



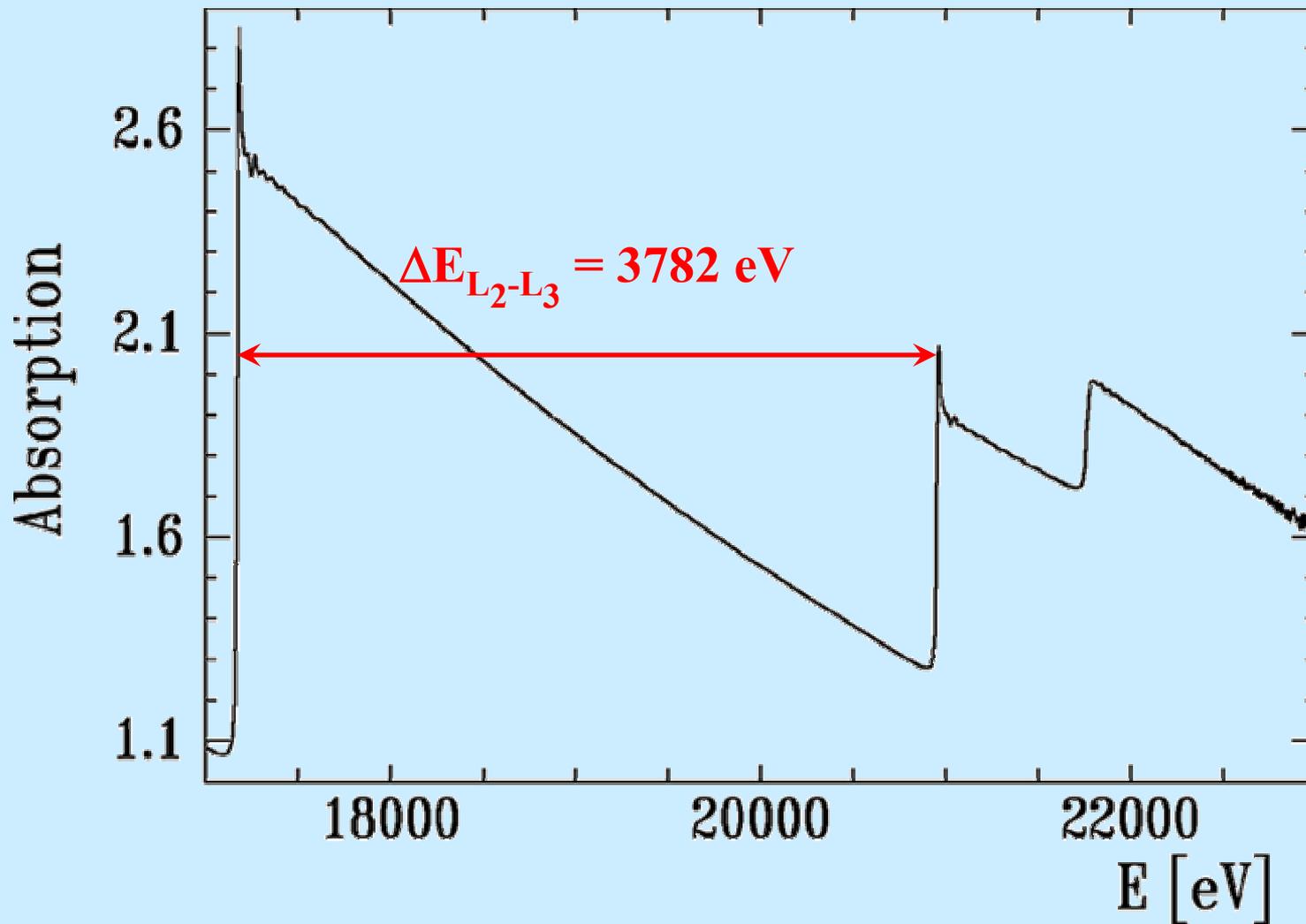
$^{74}\text{W}$	K	69 525 eV
	L <sub>1</sub>	12 100 eV
	L <sub>2</sub>	11 544 eV
	L <sub>3</sub>	10 207 eV

W powder (300 K), 40.8 s (0.02 s / point)

X-11A at the NSLS



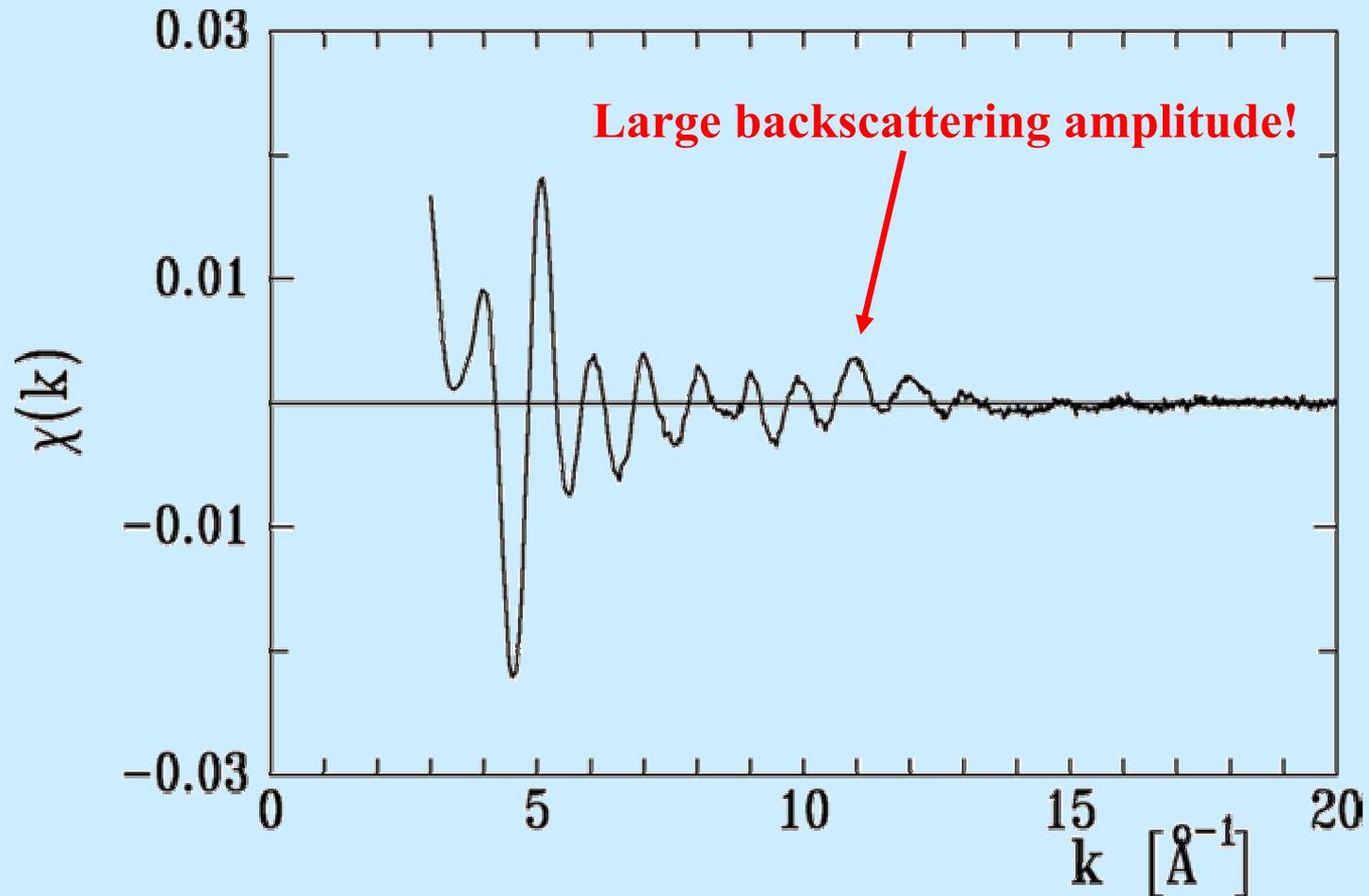
# Extreme energy range: U L-edges



${}_{92}\text{U}$	K	115 606
	L <sub>1</sub>	21 757
	L <sub>2</sub>	20 948
	L <sub>3</sub>	17 166

U L-edges, 115 s (0.05 s / point)

# Extreme energy range: U L-edges



U L<sub>3</sub>-EXAFS  
 $\Delta k = 31.5 \text{ \AA}^{-1}$  possible!



# EXAFS / XANES of heavy elements

## EXAFS:

⇒ For good EXAFS-measurements (1 keV range).  
K-edges up to  $\approx {}_{70}\text{Yb}$  ( $E_{\text{K}} = 61.332 \text{ keV}$ ) necessary.

What about heavier elements like  ${}_{92}\text{U}$ ?

L-edges overlap with K-edges of common materials,  
e.g. Pu in Zircon ( $\text{ZrSiO}_4$ ).

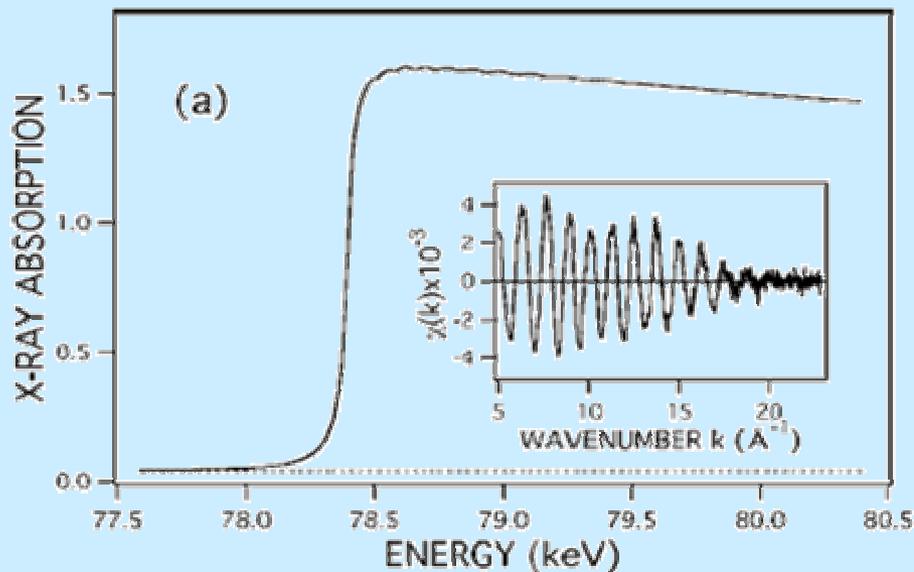
K-edges of neighbouring heavy elements do not overlap, L-edges do!  
 ${}_{92}\text{U}/{}_{91}\text{Pa}$ :  $\Delta E_{\text{K}} = 3005 \text{ eV}$ ,  $\Delta E_{\text{L}_3} = 433 \text{ eV}$ .

## XANES:

L-edges can usually be measured.

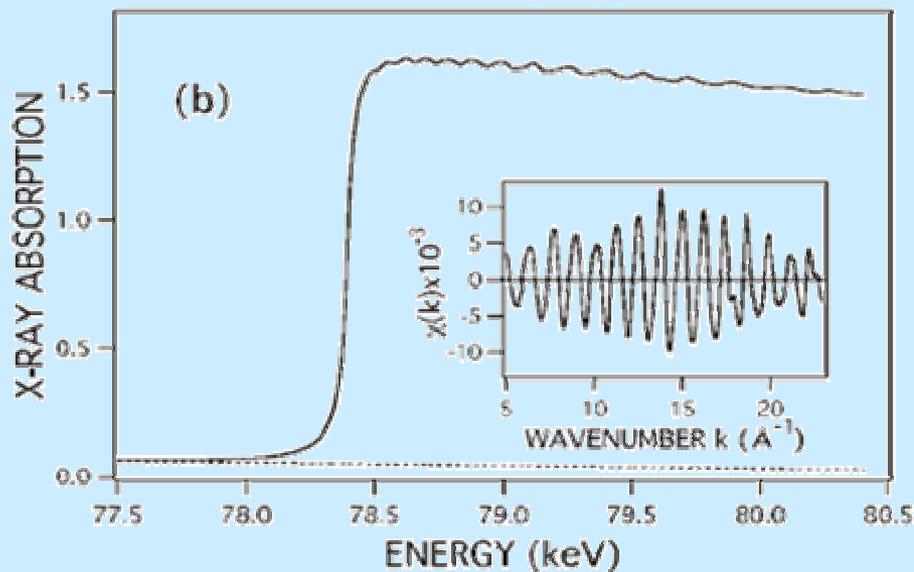


# Pt K-edge



Pt-foil

Room temperature (299 K)



12 K



# Lifetime broadening

Heisenberg's uncertainty relation:

$$\Delta t \Delta E \geq \hbar/2, \hbar = 6.6 \cdot 10^{-16} \text{ [eV}\cdot\text{s]}$$

Limited lifetime of vacancy  $\Rightarrow$  energy broadening of the edge

Modification of EXAFS-formula:

Convolution with Lorentzian function with width  $\Gamma$ :

$$\chi(E) = \chi_0(E) * g(E, \Gamma_K)$$

Finite lifetime modifies amplitudes *and* phases of the EXAFS

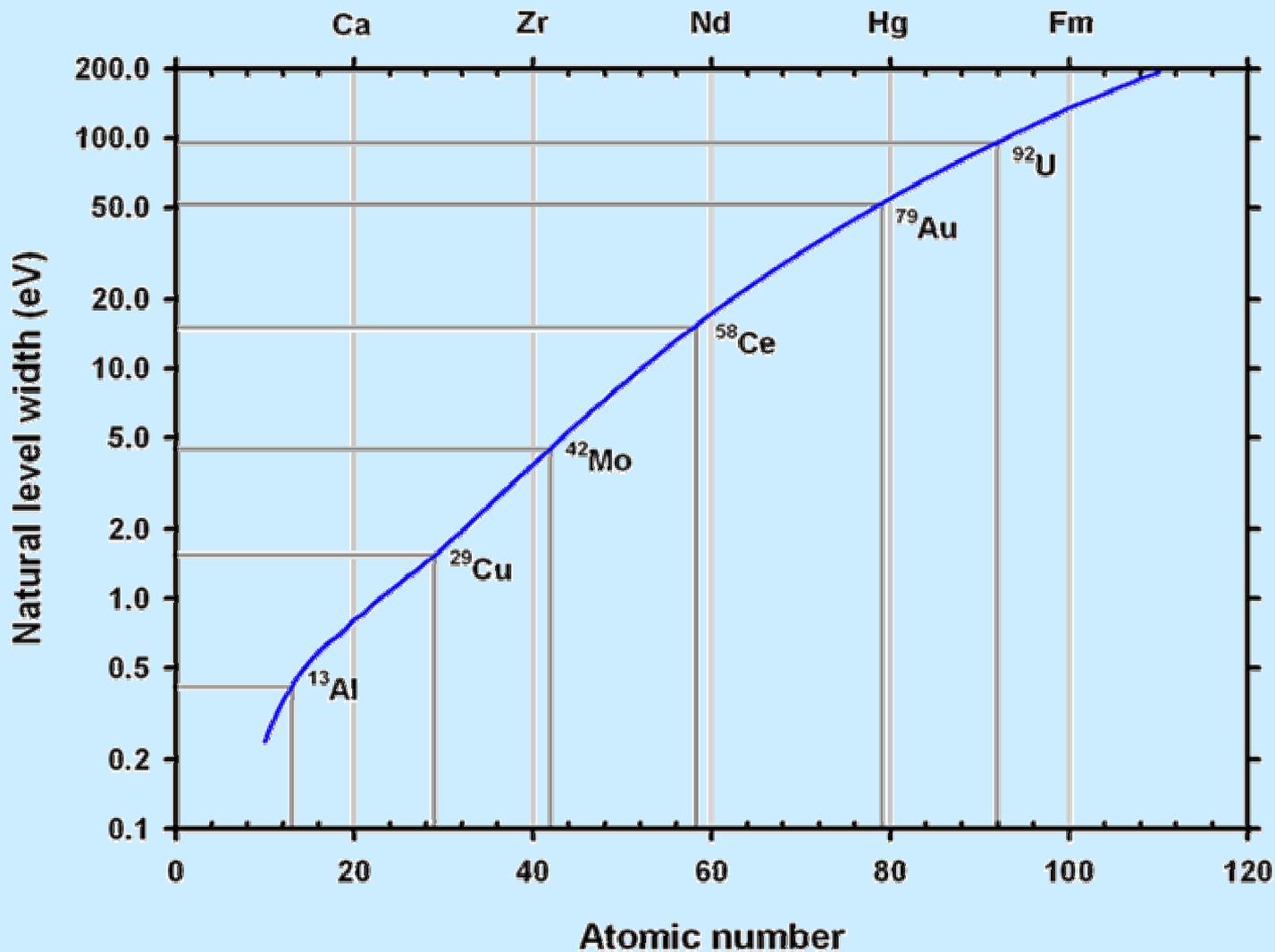
[D.G. Stearns, Phil. Mag. B 49, 541 (1984)]

Tabulations of experimental and theoretical lifetime broadening:

M.O. Krause and J.H. Oliver, J. Phys. Chem. Ref. Data 8, 329 (1979)

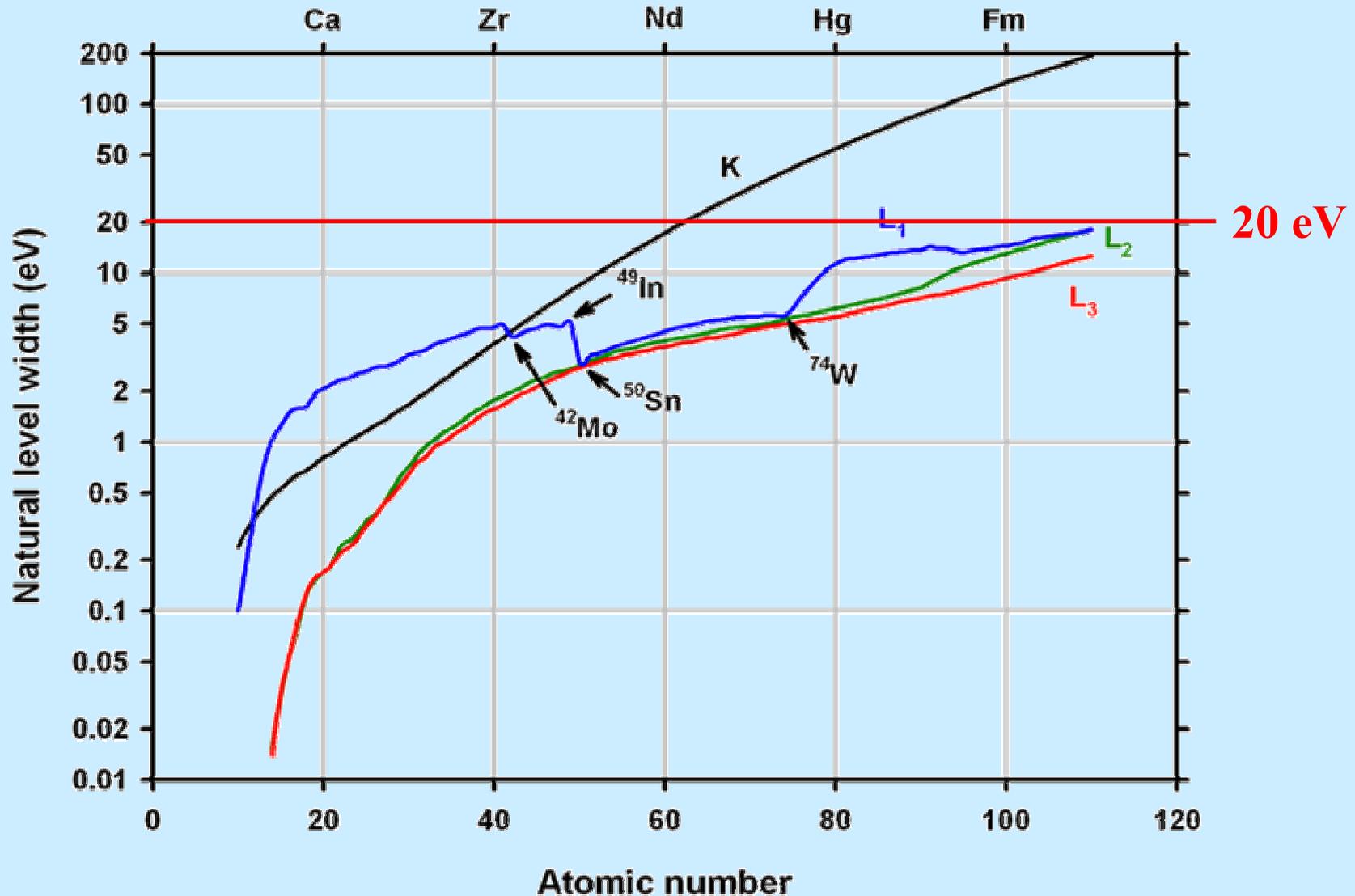


# Natural edge widths of the K-shell



M.O. Krause and J.H. Oliver, *J. Phys. Chem. Ref. Data* **8**, 329 (1979)

# Natural edge widths of K- and L-shells



M.O. Krause and J.H. Oliver, *J. Phys. Chem. Ref. Data* **8**, 329 (1979)



# Numerical deconvolution of the lifetime broadening

## Convolution theorem of Fourier-transforms:

$$FT\{f * g\} = FT(f) \cdot FT(g) \Rightarrow f = FT^{-1}\{FT(f * g) / FT(g)\}$$

## Lifetime broadening:

Convolution with Lorentzian function, FWHM  $\Gamma$ , HWHM  $\Gamma' = \Gamma/2$ .

$$\text{Lorentzian: } L(x) = \frac{1}{\pi\Gamma'} \frac{1}{1 + (x/\Gamma')^2}$$

$$\Rightarrow FT\{L(x)\} = \tilde{L}(q) = \exp(-\Gamma'|q|)$$

$\Rightarrow$  Deconvolution in Fourier space: Multiplication with  $\exp(+\Gamma'|q|)$

$\Rightarrow$  **Numerical deconvolution: Needs VERY good, low noise data.**



# Numerical deconvolution of the lifetime broadening

A. Filipponi, J. Phys. B: At. Mol. Opt. Phys. 33, 2835 (2000)

**Decomposition** of measured spectrum  $\alpha(E)$  into three components:

$$\alpha(E) = (aE + b) + S(E) + \beta(E)$$

Linear pre-edge  
background

Step function  
representing edge

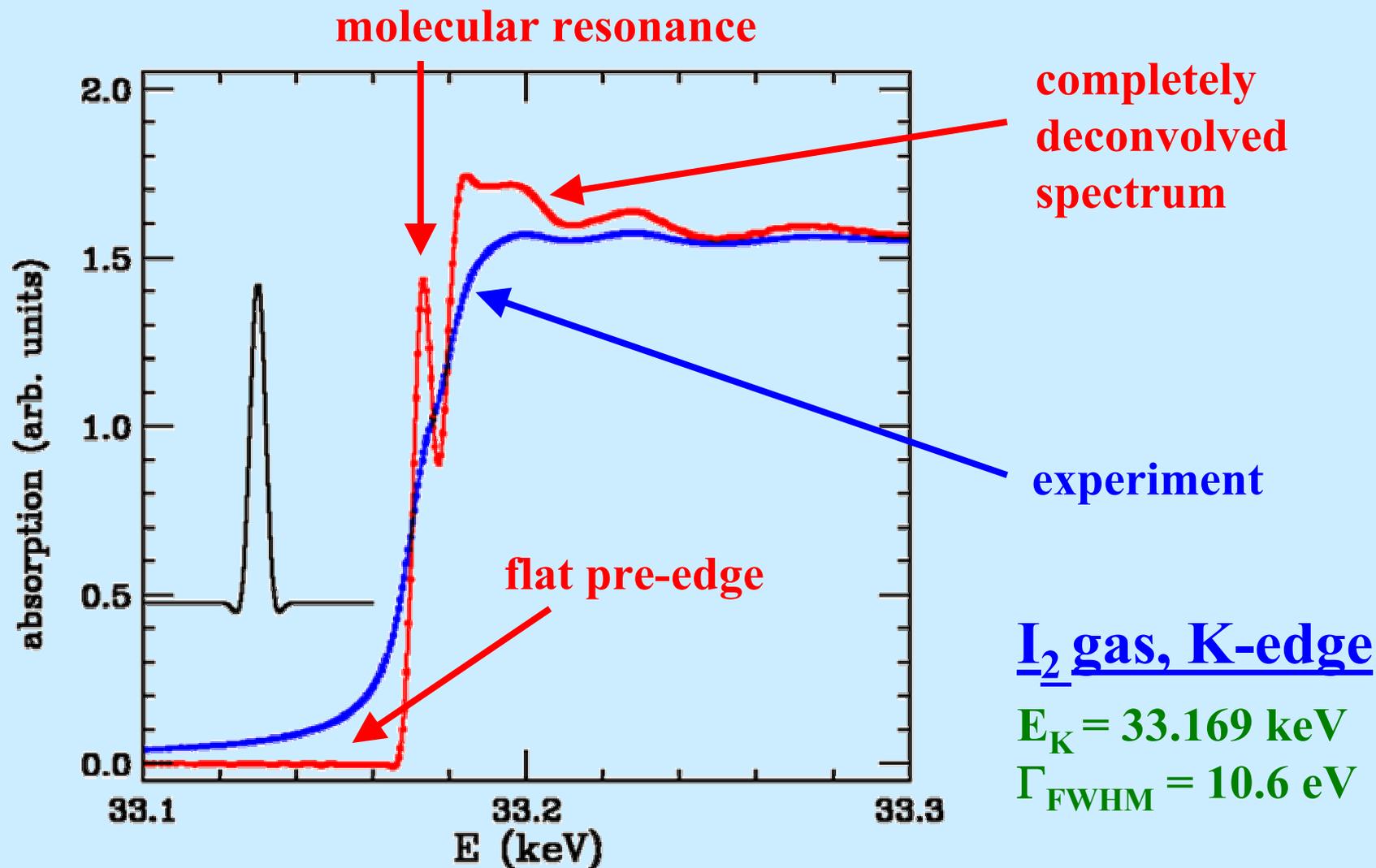
⇒ Oscillating function  
with smooth decays to 0  
at both ends

⇒ Suitable for FT

During deconvolution:

unaffected / analytically solved / numerically deconvolved

# Numerical deconvolution: I K-edge





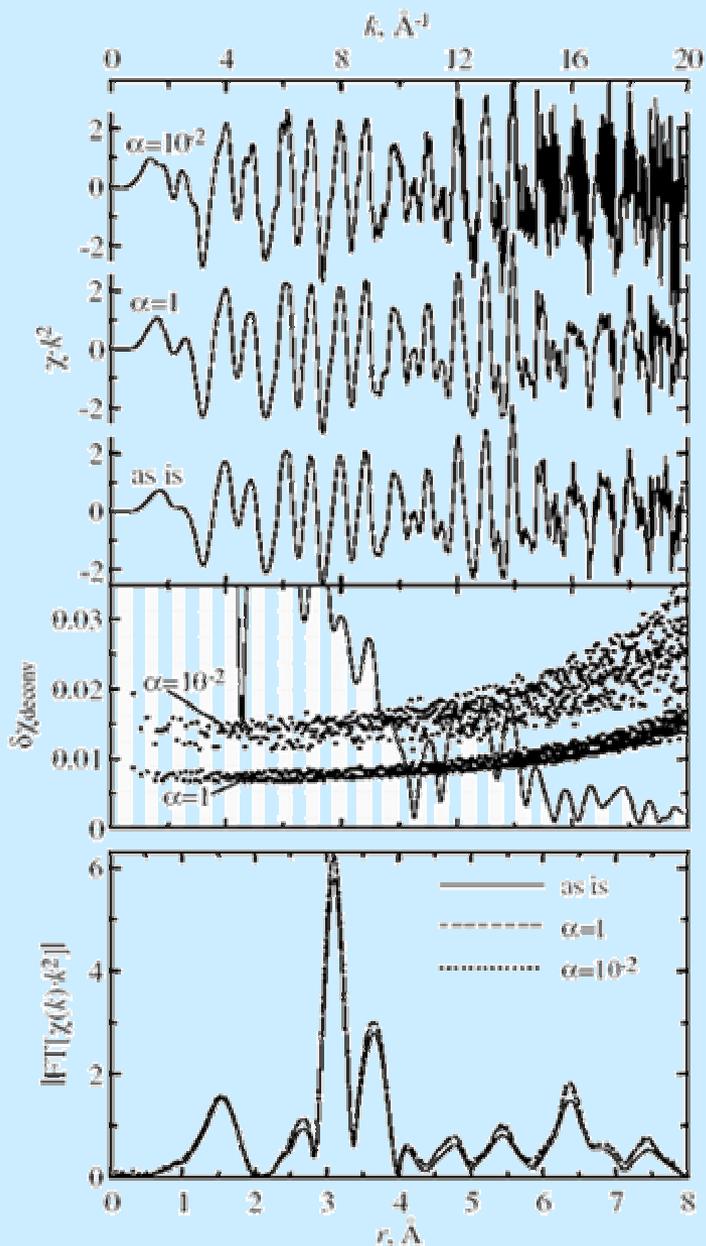
# Numerical deconvolution

K.V. Klementev, *J. Phys. D: Appl. Phys.* **34**, 2241 (2001)

In deconvolution there is usually **not** a unique solution.

However:

Klementev uses Bayesian deconvolution.  
For EXAFS: **The deconvolved results give very similar FTs**



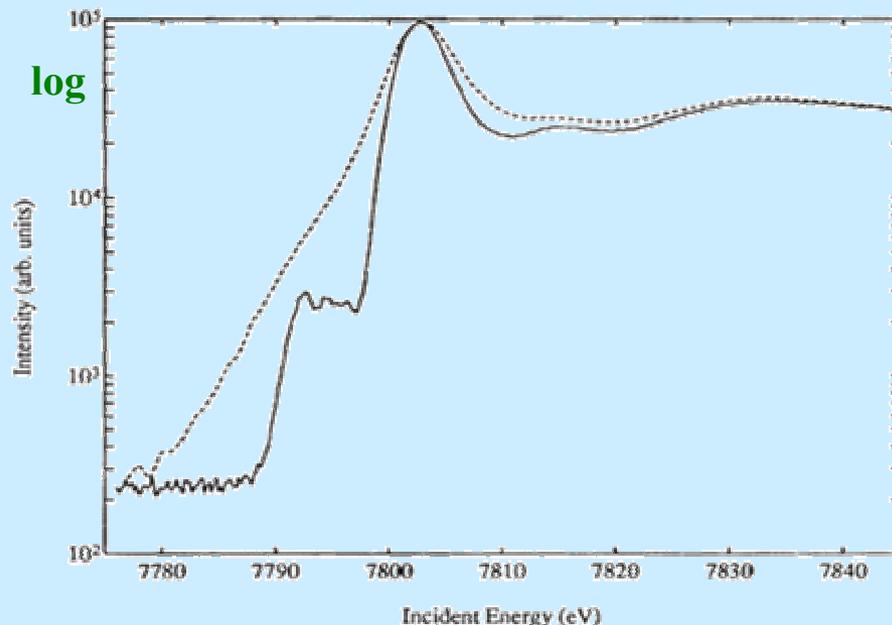
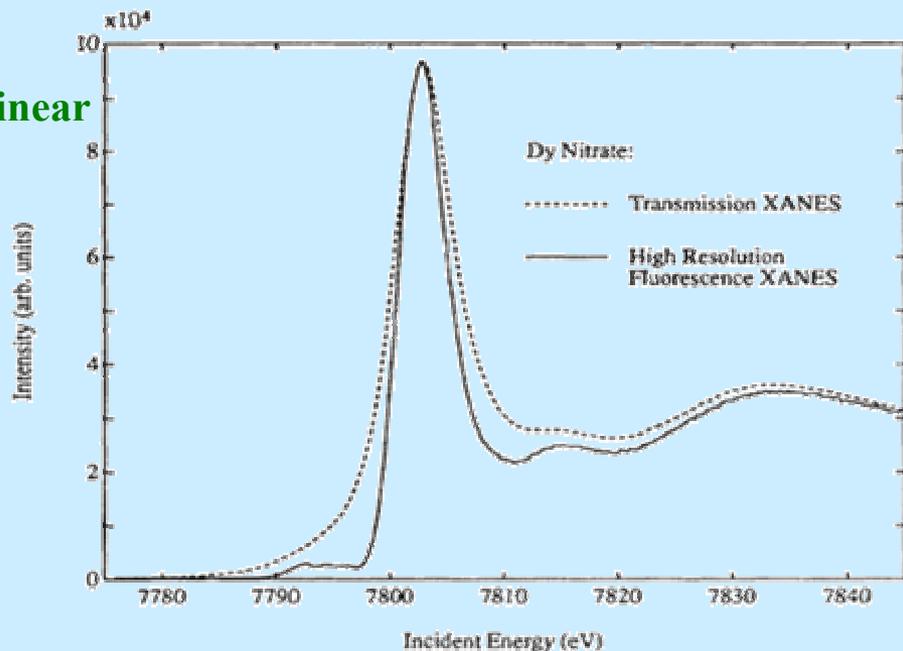


# Experimental elimination of lifetime broadening

K. Hämäläinen et al., Phys. Rev. Lett. 67, 2850 (1991)

## Prerequisites:

- High resolution monochromator ( $<\Gamma$ ), here: Si(220): 0,7 eV.
- Sharp secondary channel (fluorescence, Auger) + high resolution analyser ( $<\Gamma$ ), here: Si(440), 0.3 eV.

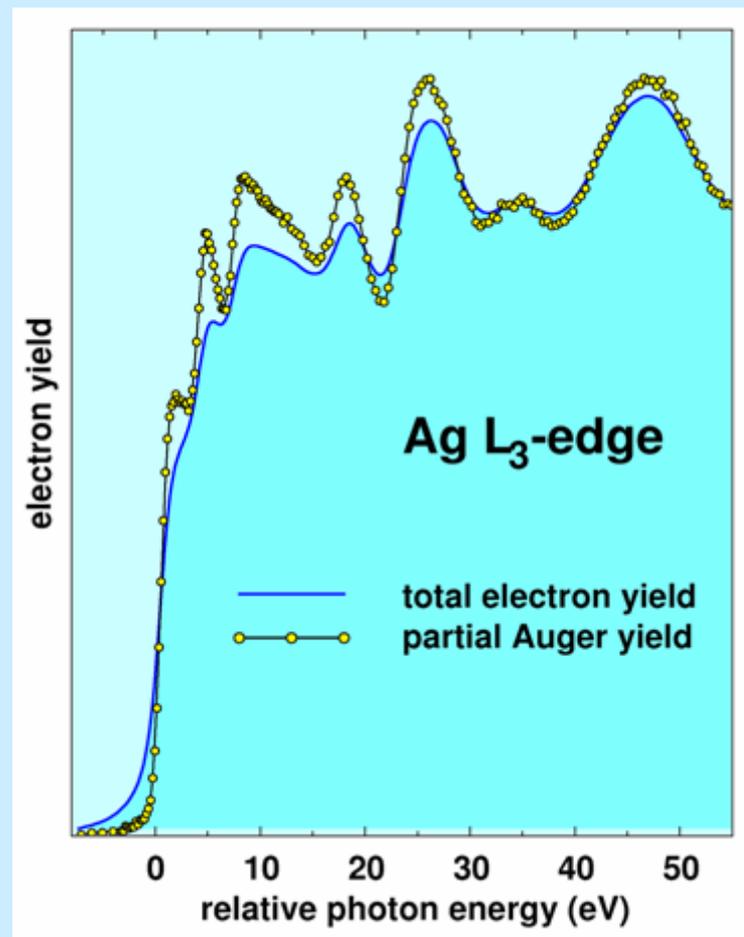
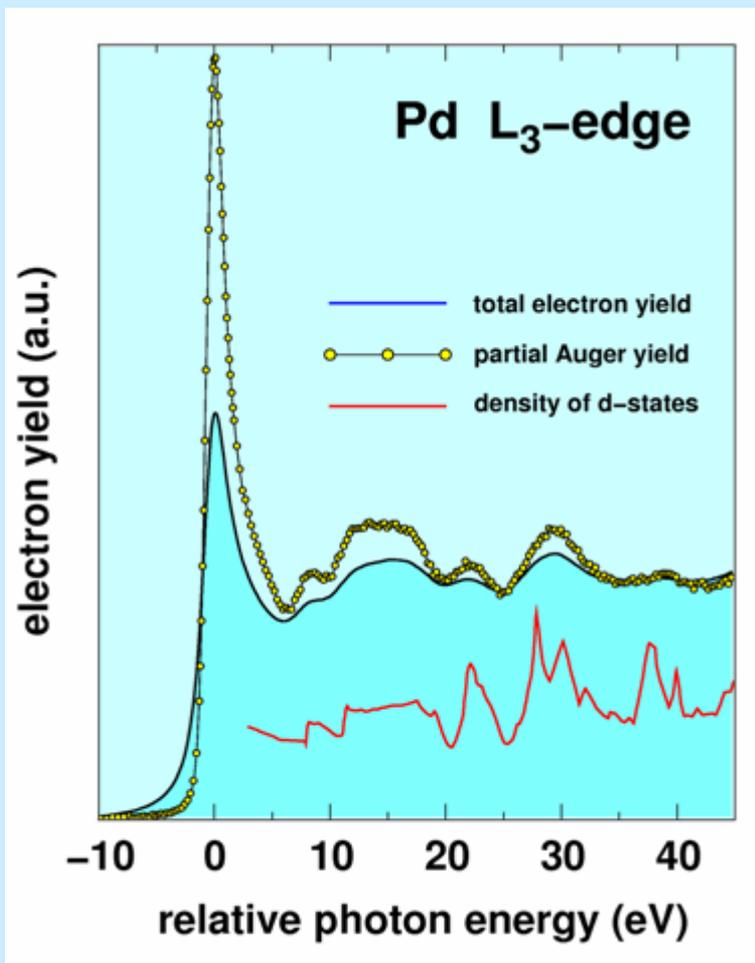


## ${}_{66}\text{Dy L}_3$ -edge in $\text{Dy}(\text{NO}_3)_3$

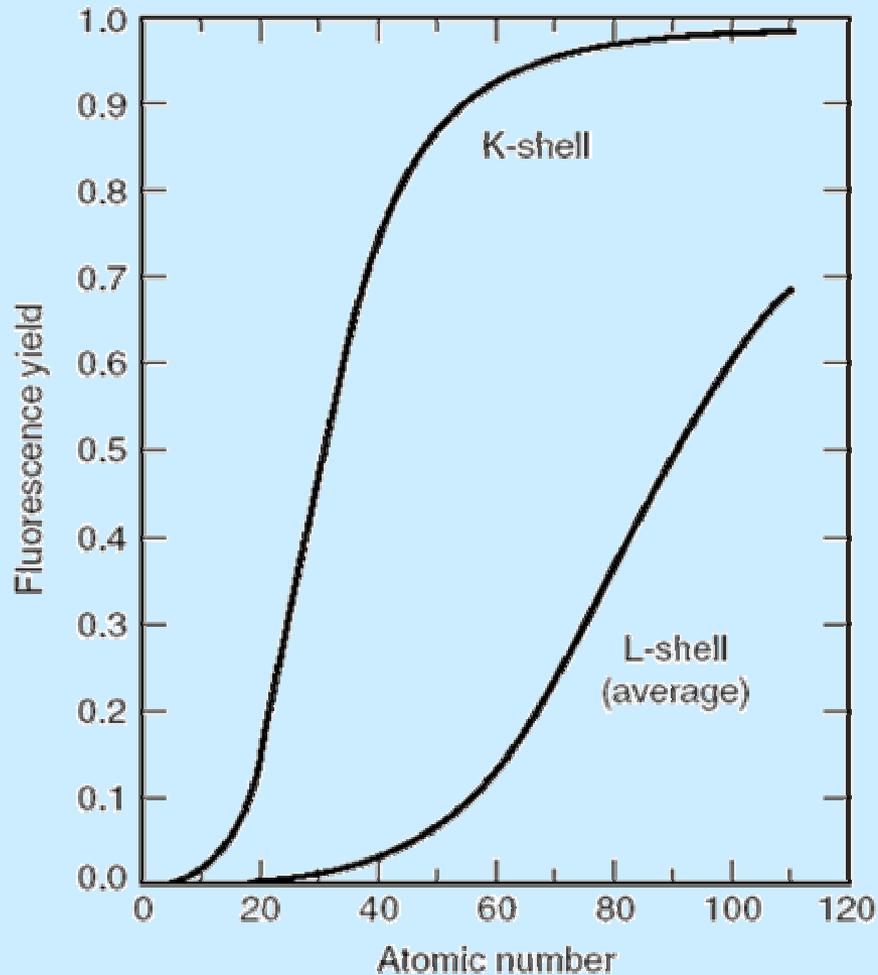
$\Gamma_{L_3} = 4.17$  eV,  $L_{\alpha 1}$ -line:  $E_{M_5} - E_{L_3} = 6495$  eV,  $\Gamma_{M_5} = 1.4$  eV

# Constant final state X-ray absorption: Sub-lifetime near edge structure

W. Drube, HASYLAB



# Fluorescence yields for K- and L-shells



**Probability of radiative processes increases with atomic number:**

**> 80% from  $_{45}\text{Rh}$**

**> 90% from  $_{56}\text{Ba}$**



# Conclusions: Removing the lifetime effect

## First step:

Experimental “sharpening” via intermediate states:

Based on narrower intermediate states with longer lifetimes ( $L_{2,3}$  for K-absorption,  $M_{4,5}$  for  $L_3$ -absorption etc.) than the hole level.

They differ by a factor of 3 to 8 to K-levels.

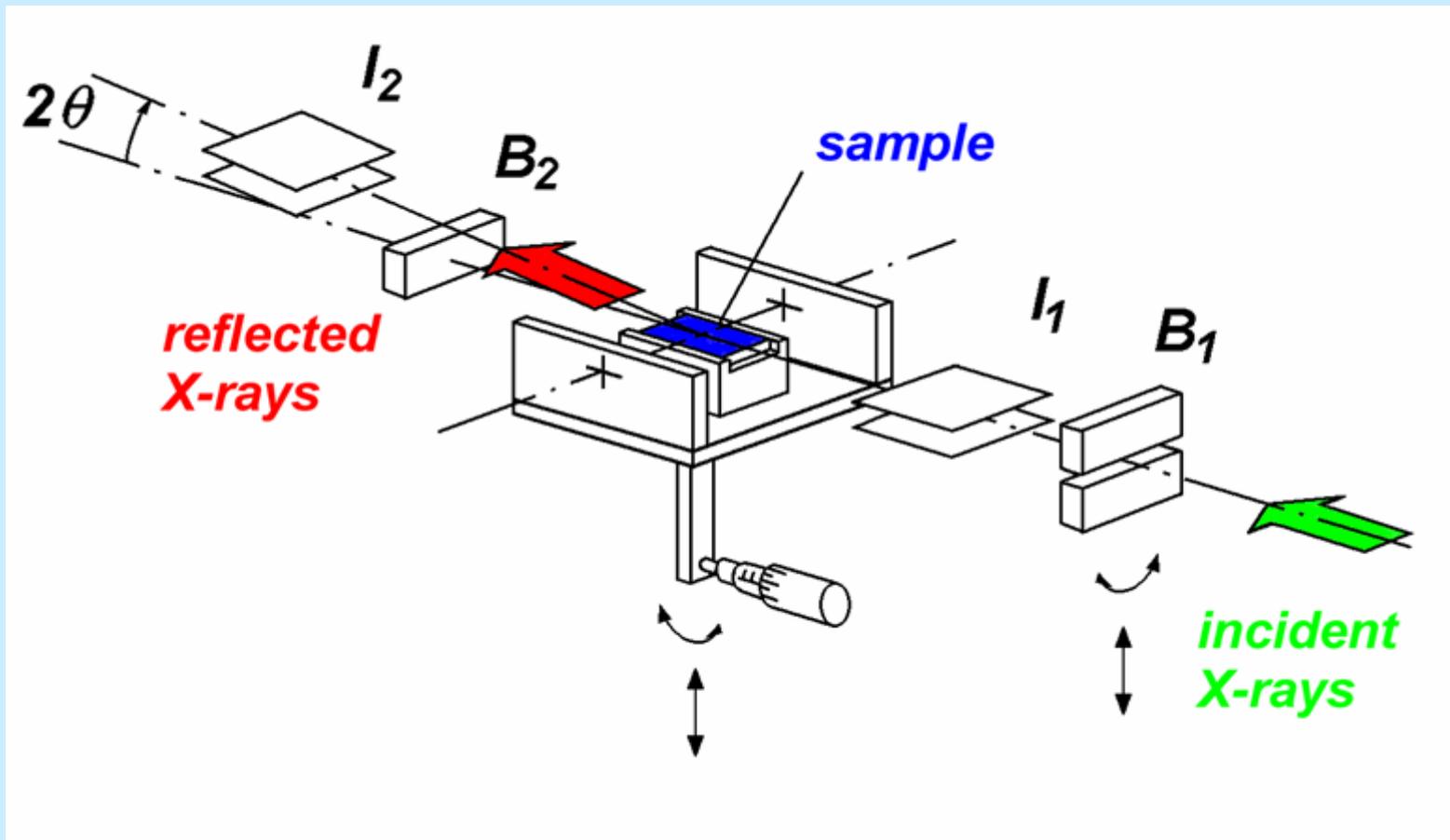
⇒ In principle the edge width can be reduced down to the width of the intermediate state,  
e.g. for the Pt K-edge from about 49eV down to 6eV.

## Second step: Numerical deconvolution of remaining width.

## However: High resolution analyser necessary!

The same is true for resonant inelastic X-ray scattering (RIXS) methods.

# EXAFS measurements in reflectivity



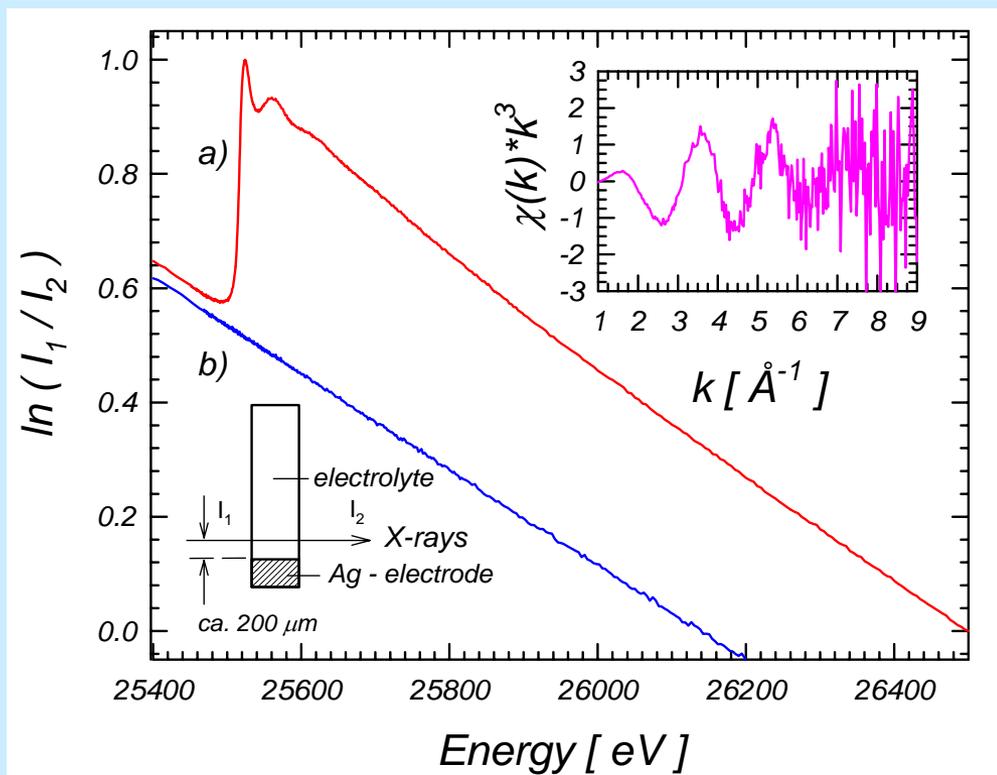
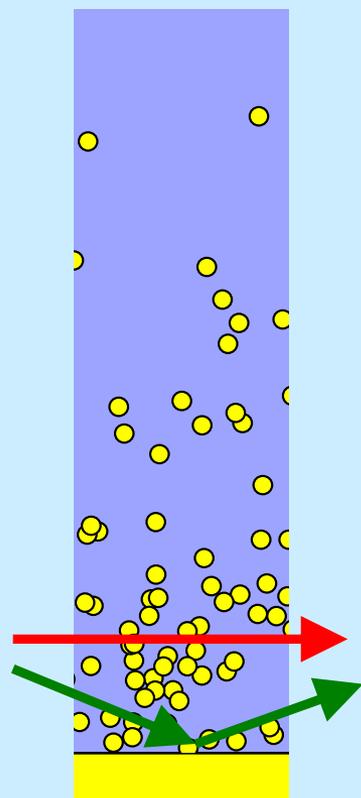
## Experiment

Intrinsic surface sensitivity, depth profiling possible



# Corrosion processes: Ag in Na<sub>2</sub>SO<sub>4</sub> (pH 6.5)

- Material is released to the liquid phase.



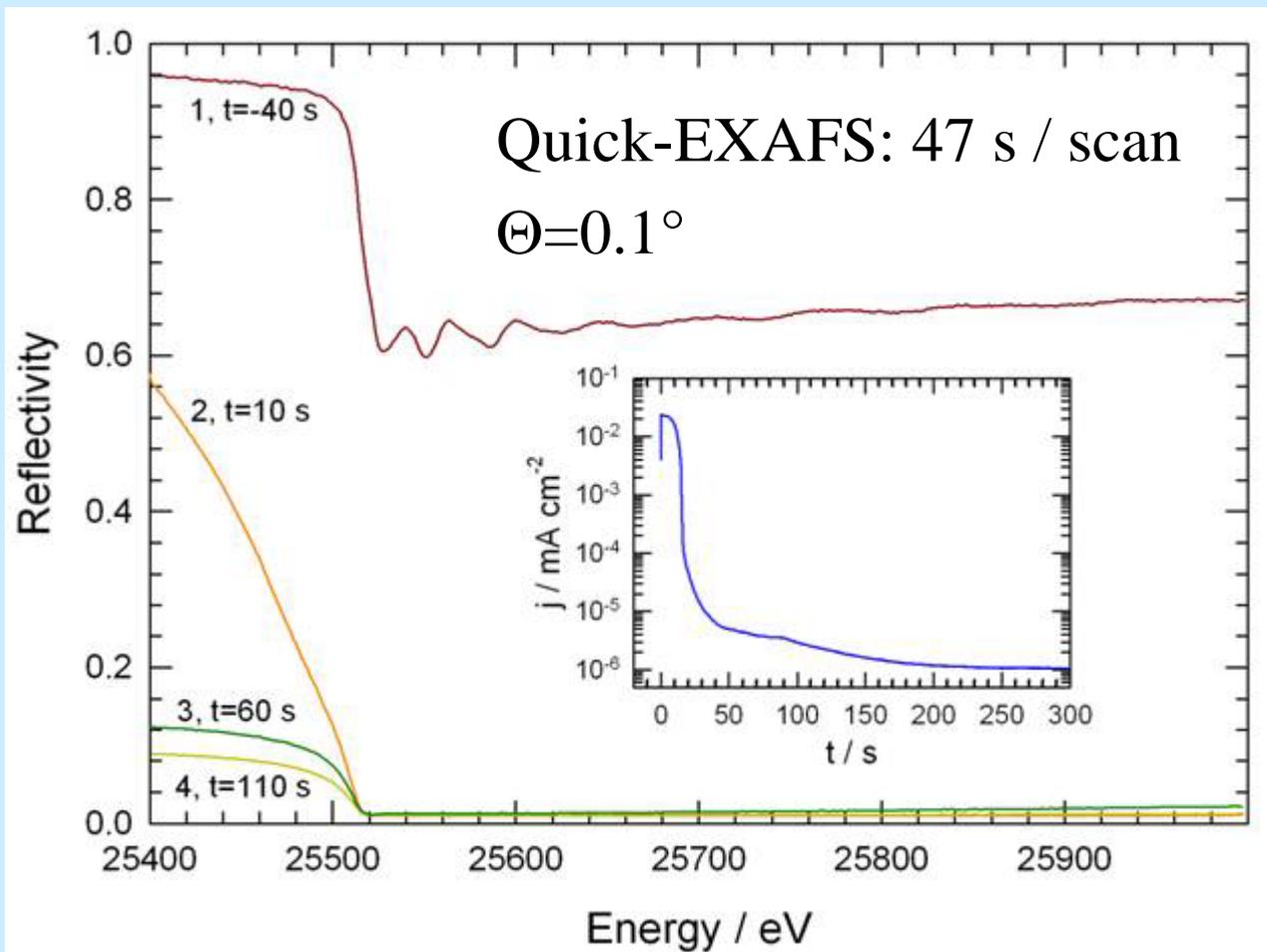
**a) 0.5M Na<sub>2</sub>SO<sub>4</sub>**  
**U=+1.20 V**

**b) 1M NaOH**  
**U=+1.1 V**

- Formation of  $\text{Ag}(\text{OH})_2^-$  in solution
- State of the surface?
- Formation of (partly) passive layers?



# Corrosion processes: Ag in Na<sub>2</sub>SO<sub>4</sub> (pH 6.5)

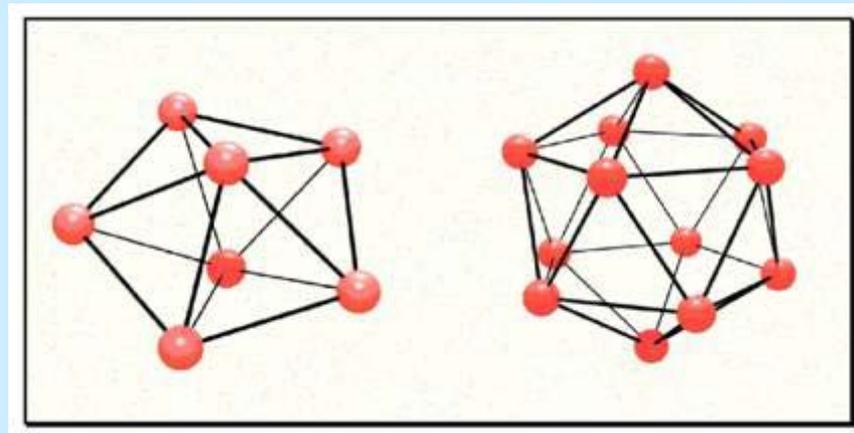


**Dramatic reflectivity changes!**



# High energy XRD

## 5-fold symmetries in liquids (e.g. liquid Pb)



Contact of liquid with crystalline surface, e.g. liquid Pb / Si (100):

Local structure at the „buried“ interface?

Hard to investigate with conventional techniques:

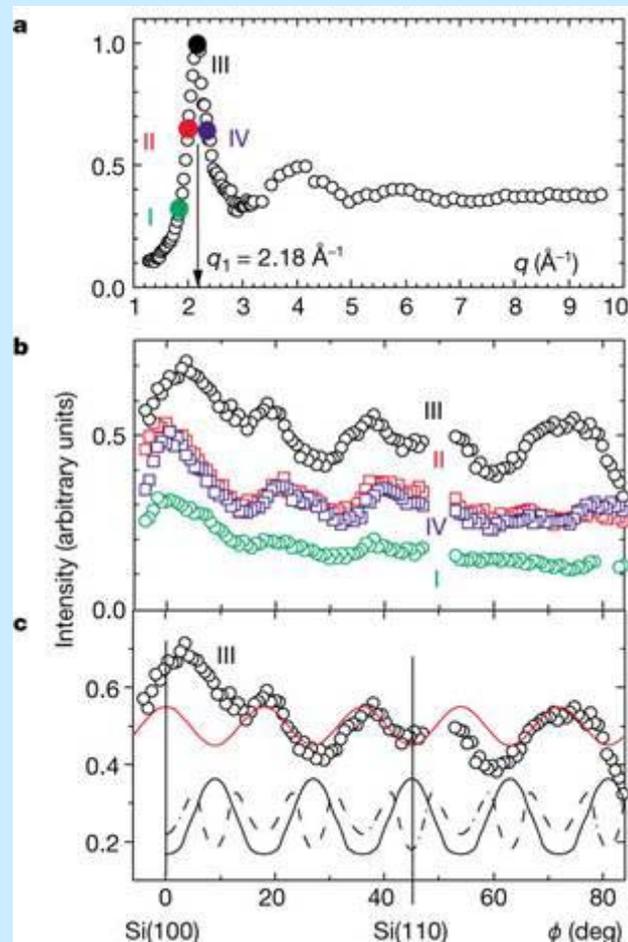
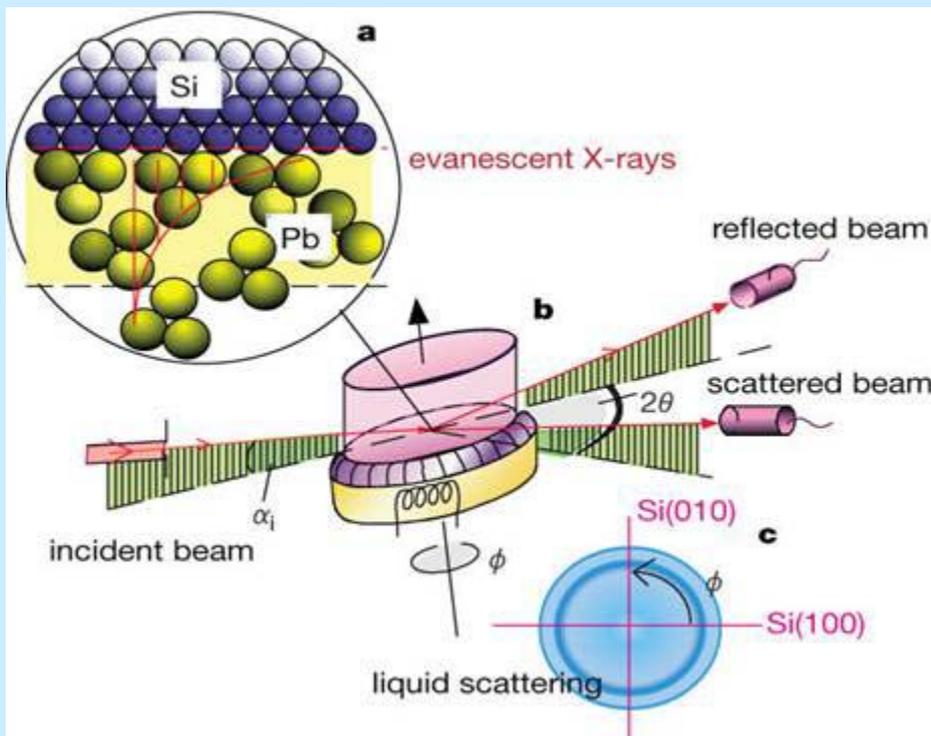
Strong „parasitic“ absorption in the liquid metal and the crystal!



# Observation of five-fold local symmetry in liquid lead

Hard X-rays: Large penetration depth!  
(Penetration of several cm of material)

collimated beam  
(10  $\mu\text{m}$  vert.)  
 $\alpha_i = 0.03^\circ$



Surface X-ray scattering experiments with high energy (70 keV)

H. Reichert et al., Nature 408 (2000) 839



# High energy XAFS

**Grazing incidence X-ray absorption spectroscopy  
at the Pb K-edge (88 keV):**

- penetration of the (optical thin) Si-material
- specular reflection at the Pb (liquid) / Si interface
- angle variation allows  
depth profiling of the local structures in the liquid
- temperature variation: surface melting of Pb/Si?

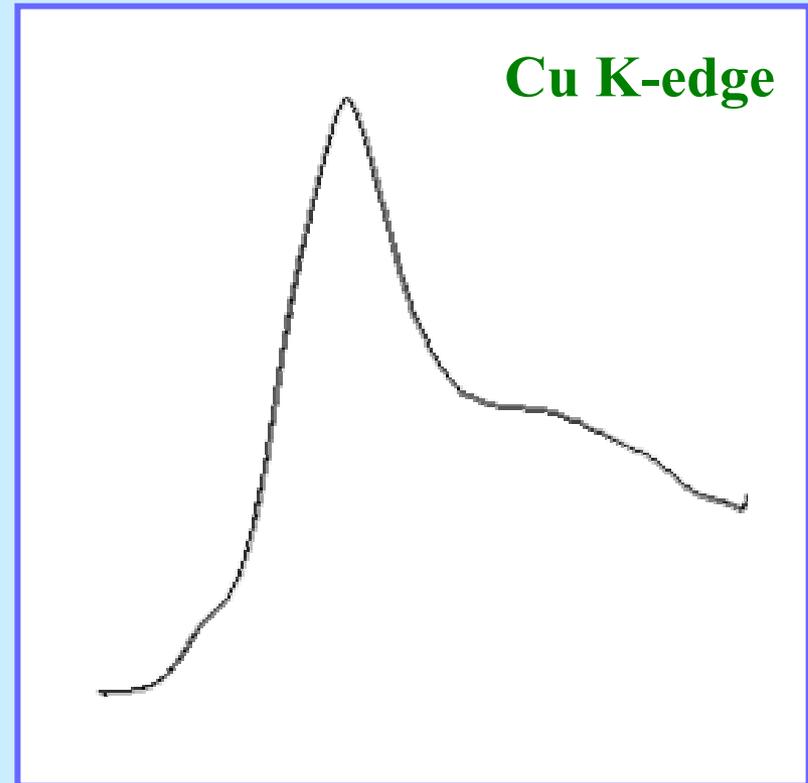


# Activation of a $\text{CuO}/\text{ZnO}/\text{Al}_2\text{O}_3$ catalyst for methanol synthesis

## Goal:

Optimization of the properties of catalysts

**At higher energies (Pt K-edge) one can look into e.g. stainless steel cells.**



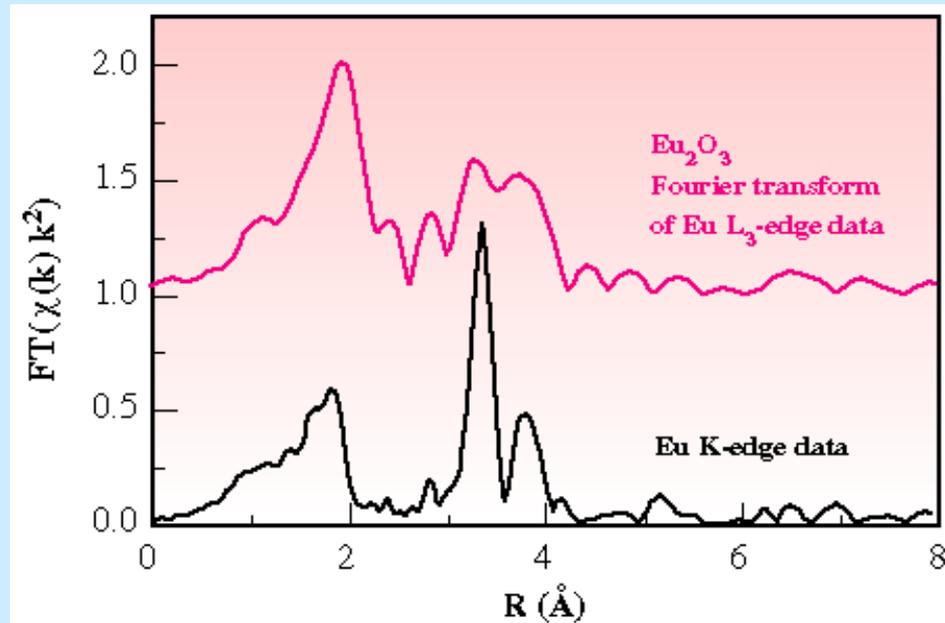
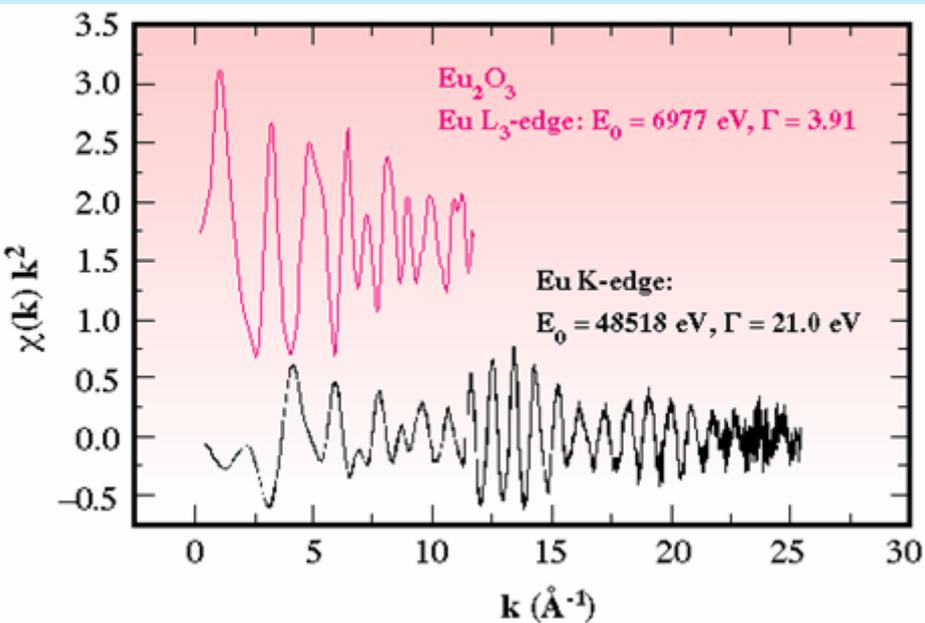
In situ reduction in  $\text{H}_2$  atmosphere

Time resolution: **50 ms**



# High energy absorption edges

Borowski, D.T. Bowron, S. de Panfilis, J. Synchrotron Rad. 6 (1999) 179



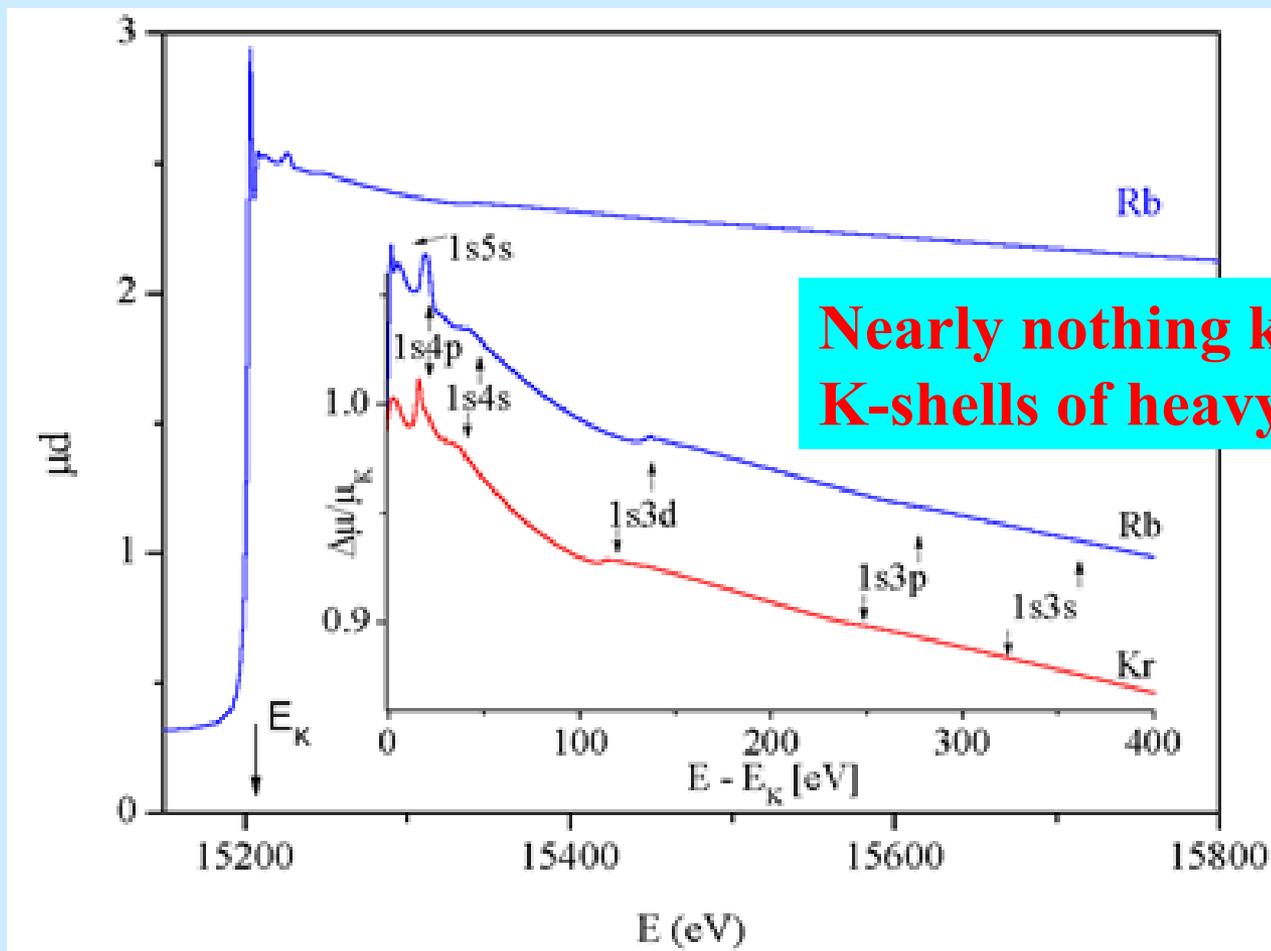
Eu<sub>2</sub>O<sub>3</sub>-Probe: Eu L<sub>3</sub>-Kante (6977 eV) + Eu K-Kante (48518 eV)

<sup>63</sup>Eu K 48518 eV  
**K-edge data are MUCH better than L<sub>3</sub>-edge**  
despite lifetime broadening!  
L<sub>3</sub> 6977 eV ⇒ ΔE(L<sub>3</sub>-L<sub>2</sub>) = 640 eV



# Multishell excitations: Rb und Kr

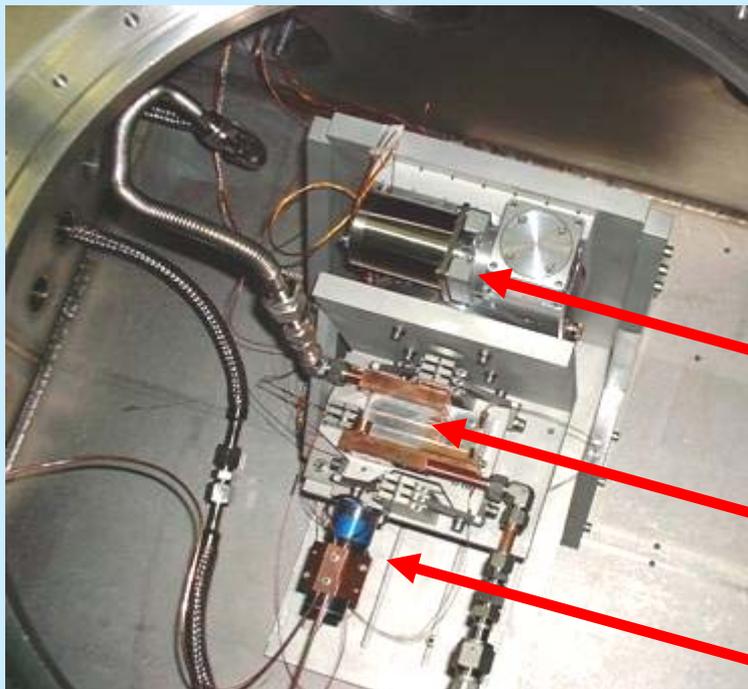
Rb+Kr benachbarte Elemente



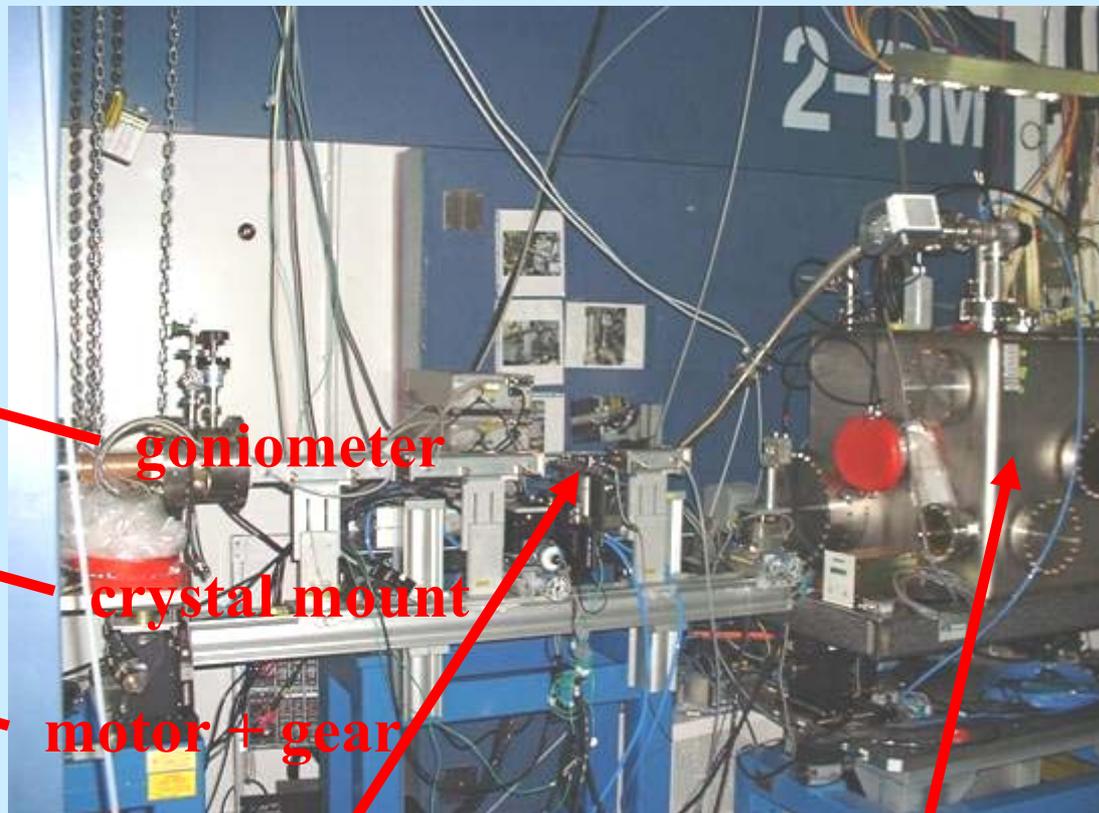
Nearly nothing known about  
K-shells of heavy elements



# QEXAFS Experiments at the APS undulator 1-ID



Monochromator  
(crystal not mounted),  
cryogenic cooling



goniometer

crystal mount

motor + gear

Experimental setup

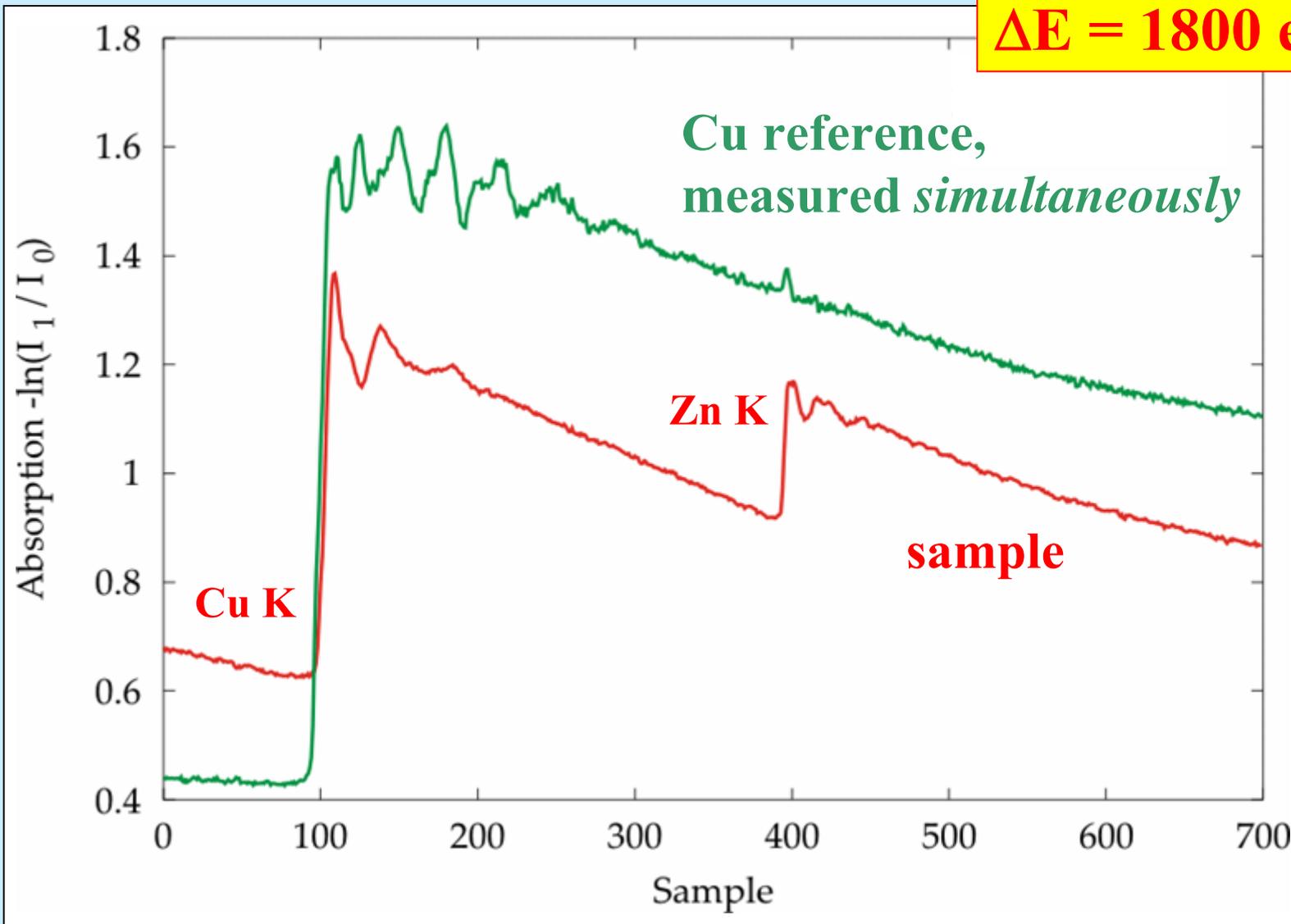
sample

monochromator



# CuO/ZnO-catalyst: Single scan, 50 ms

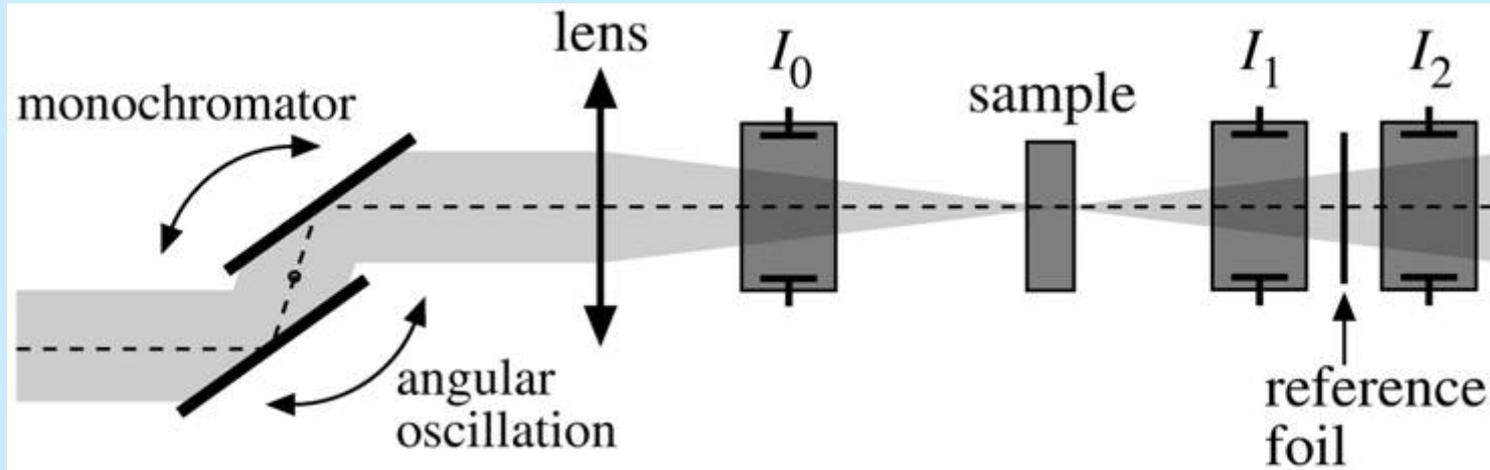
**$\Delta E = 1800 \text{ eV!}$**



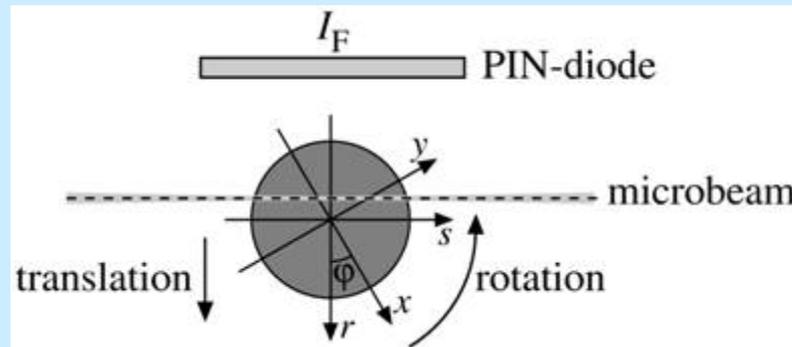


# XANES-Tomography

Goal: 3-d Imaging with microscopic resolution

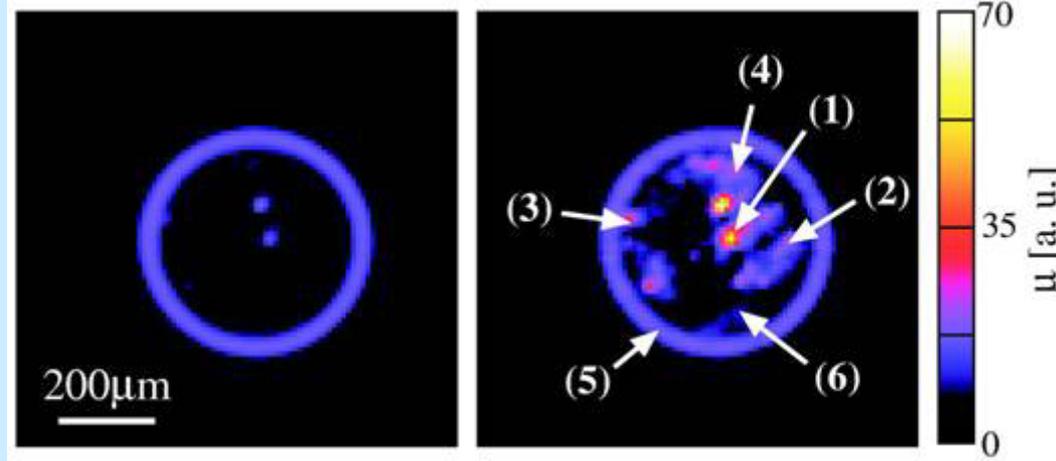


Experimental setup, side view

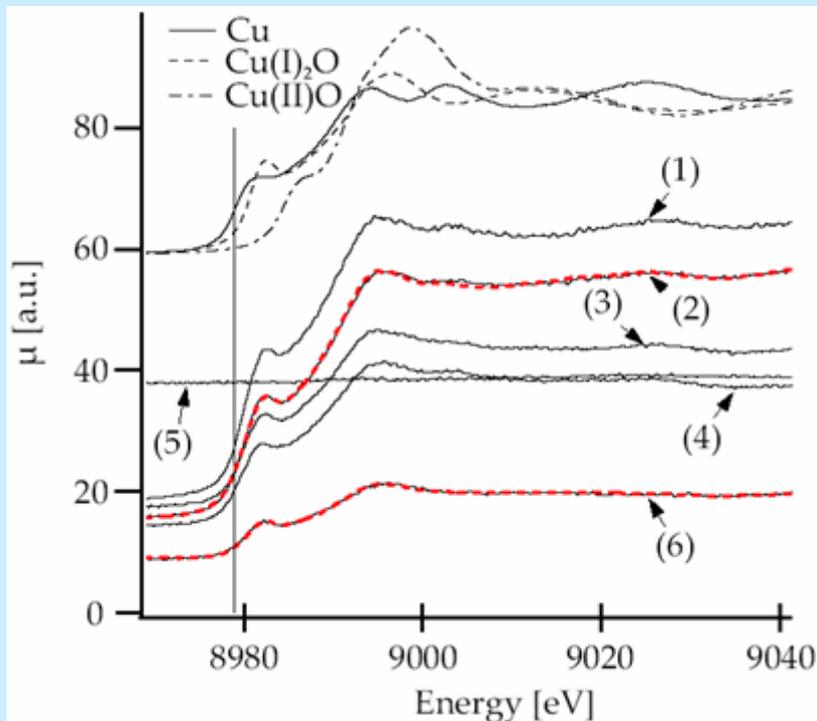


Sample stage, top view

# Cu/ZnO catalyst



## Sample below / above Cu K-edge



Reconstruction at different positions  
after several oxidation/reduction cycles

Sample in glass capillary, outer  
diameter 500 mm, inner diameter 400  
mm.

Beam size: 10 μm x 10 μm

1s/sample position, 10 Hz

(Appl. Phys. Lett. 2004)

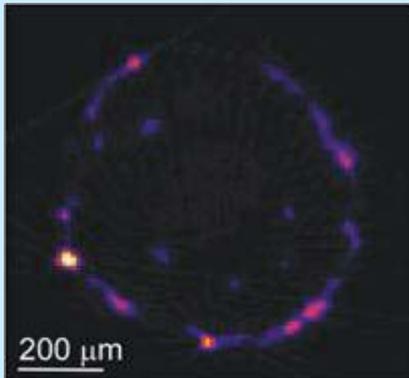
## Reconstructed Spectra with reference compounds



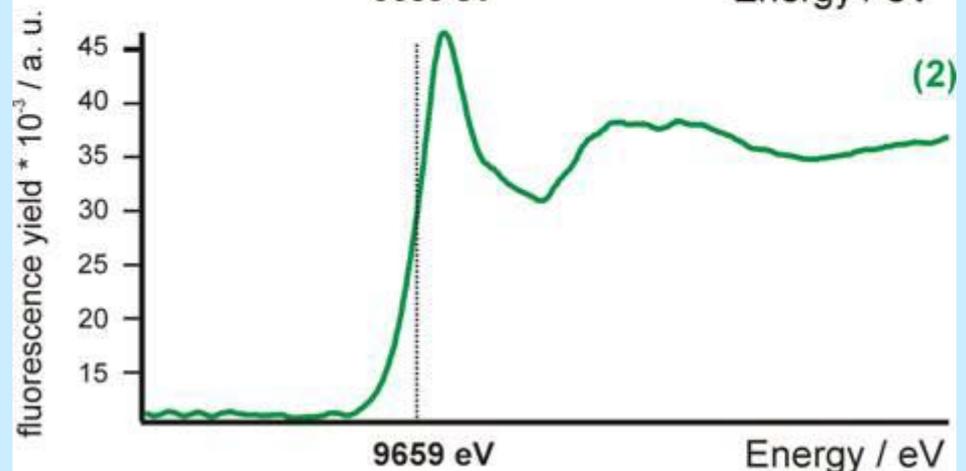
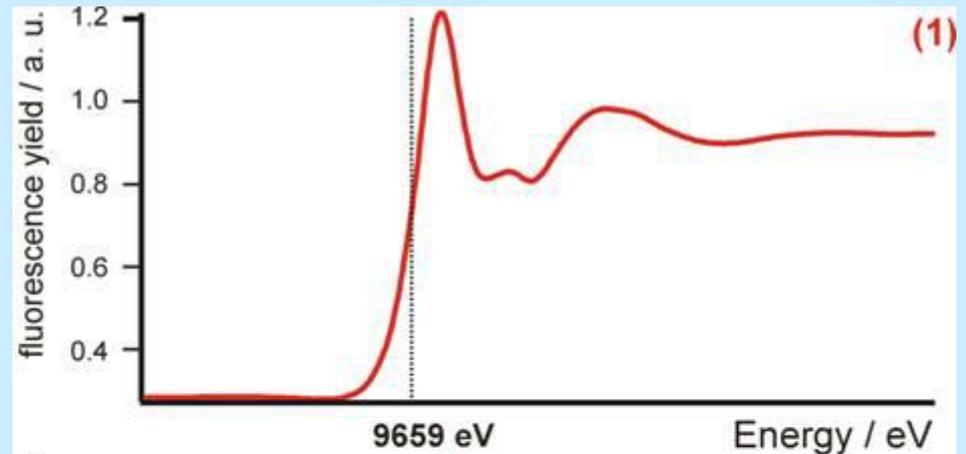
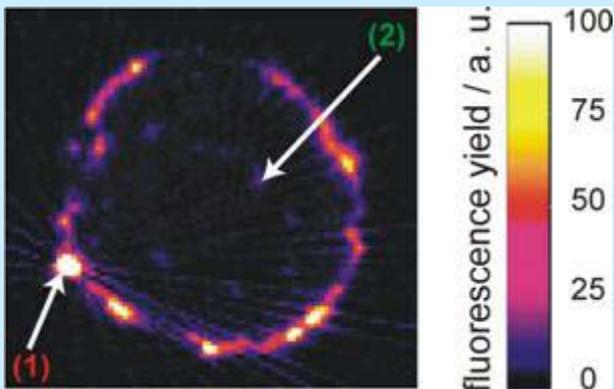
# Tomato root grown on a polluted (Zn, Pb) soil

- Low metal-ion concentration (<100 ppm):  
→ Fluorescence tomography at the **Zn K-edge**

## Below Zn K-edge



## Above Zn K-edge





# $\mu$ -XAFS at high energies

## Advantage:

High penetration through sample cells and other materials, e.g.  
characterization of multicomponent samples on the  $\mu\text{m}/\text{nm}$ -scale.

**$\Rightarrow$  Valence distribution of elements**



# Experiments at PETRA III

## PETRA III: A Low Emittance Synchrotron Radiation Source

### Technical Design Report February 29, 2004

(download from [www.desy.de](http://www.desy.de))

**“The construction work will start in July 2007.  
First photons for user experiments are expected in 2009.”**

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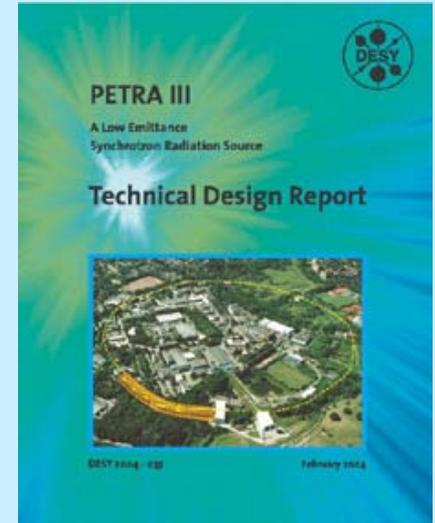
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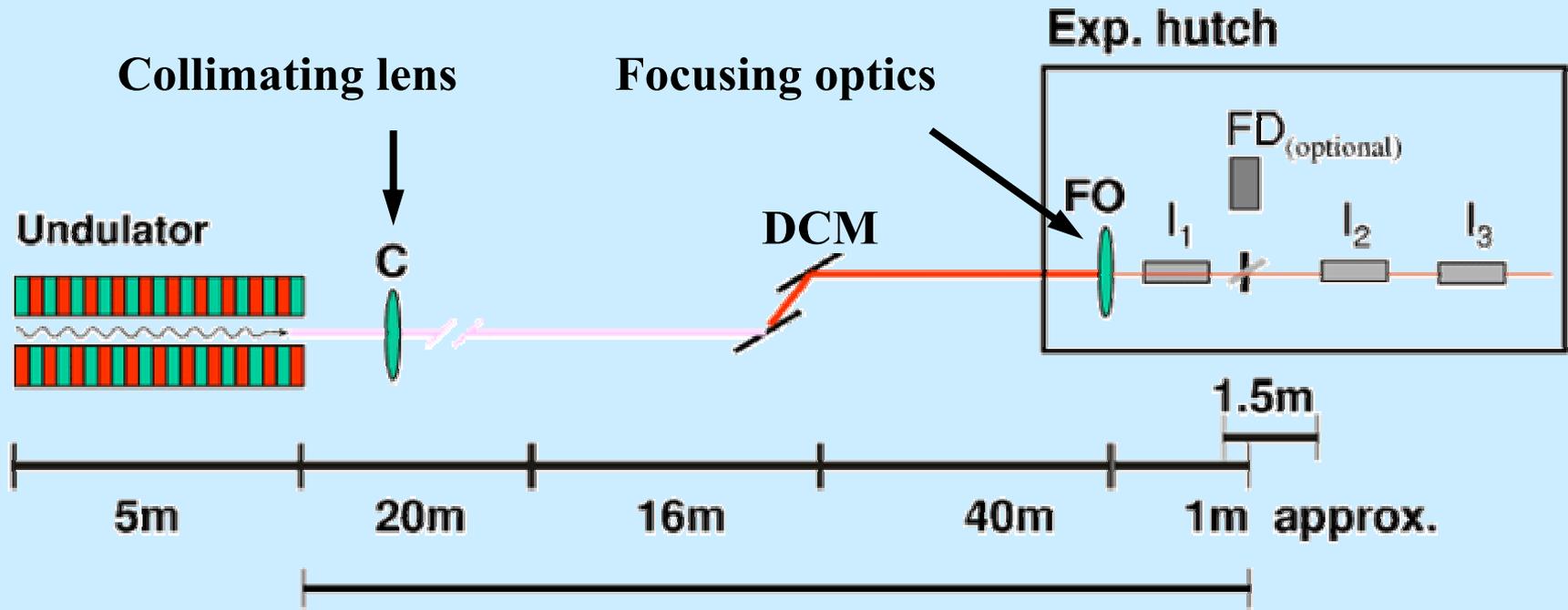
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# Beamline layout at PETRA III



## Absorption beamline at PETRA III for $E = 30 - 100$ keV

Si(311), Si(511)-crystals,  
good energy resolution achievable (few eV).

Reference:





# Beamline requirements

## Monochromator:

Fast scans for time resolved studies (500 eV/s, basic option)

## Insertion device:

Undulator with taper-option **necessary** to give 1500 eV bandpass

## If highest brilliance necessary:

Undulator scanned with monochromator (500 eV/s)



# High energy absorption edges

- K-edges: Larger wavenumbers compared to L-edges possible

- K-edges thus more sensitive for disorder

Theory: Van Hung, Ba Duc, Frahm: J. Phys. Soc. Japan 72, 1254 (2003)

- Good penetration of sample cells

(several mm of metals like Fe, several cm of Si or H<sub>2</sub>O)

- But: Larger lifetime broadening of K-XAFS

⇒ New methods necessary (numerical / experimental)

However: Lifetime broadening seems to be not as critical as expected, already very promising results in literature!

The high energy range offers new scientific possibilities for XAS!



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**RWTH Aachen: B. Lengeler, C. Schroer**

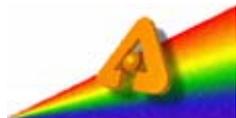


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